

## Memory-Enhanced Plasticity Modeling of Sand Behavior under Undrained Cyclic Loading

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1	Memory-enhanced plasticity modelling of sand behaviour under undrained
2	cyclic loading
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#### 11 ABSTRACT

This work presents a critical state plasticity model for predicting the response of sands to cyclic 12 loading. The well-known bounding surface SANISAND framework by Dafalias and Manzari (2004) 13 is enhanced with a 'memory surface' to capture micro-mechanical, fabric-related processes directly 14 effecting cyclic sand behaviour. The resulting model, SANISAND-MS, was recently proposed by 15 Liu et al. (2019), and successfully applied to the simulation of drained sand ratcheting under thou-16 sands of loading cycles. Herein, novel ingredients are embedded into Liu et al. (2019)'s formulation 17 to better capture the effects of fabric evolution history on sand stiffness and dilatancy. The new 18 features enable remarkable accuracy in simulating undrained pore pressure build-up and cyclic 19 mobility behaviour in medium-dense/dense sand. The performance of the upgraded SANISAND-20 MS is validated against experimental test results from the literature — including undrained cyclic 21 triaxial tests at varying cyclic loading conditions and pre-cyclic consolidation histories. The pro-22 posed modelling platform will positively impact the study of relevant cyclic/dynamic problems, for 23 instance, in the fields of earthquake and offshore geotechnics. 24

#### 26 INTRODUCTION

Geotechnical structures subjected to cyclic loading may experience severe damage, or even 27 failure, due to the soil losing its shear strength and stiffness, or experiencing excessive deformation 28 under numerous loading cycles (Andersen 2009). Sound engineering analysis of these geotechnical 29 systems must rely on accurate simulation of cyclic soil behaviour. This is to be pursued by means 30 of constitutive models capable of reproducing a number of fundamental features of soil response 31 under cyclic loading, such as irreversible/plastic straining (Youd 1993; Vaid and Thomas 1995), 32 cyclic hysteresis (Berrill and Davis 1985; Kokusho 2013) and pore water pressure build-up (Seed 33 and Rahman 1978; Berrill and Davis 1985; Ishihara 1993; Kokusho 2013) under a wide range of 34 initial/boundary/drainage conditions. 35

In the past decades, a plethora of constitutive models – from very simple to highly sophisticated 36 - have been proposed to reproduce cyclic soil behaviour in engineering applications. The case 37 of sandy soils attracted particular attention after catastrophic geotechnical failures during seismic 38 events (Ishihara 1993). The families of multi-surface (Prévost 1985; Elgamal et al. 2003; Houlsby 39 and Mortara 2004) and bounding-surface (Dafalias and Popov 1975; Manzari and Dafalias 1997; 40 Papadimitriou and Bouckovalas 2002; Pisanò and Jeremić 2014) plasticity models have proven 41 successful in capturing relevant features of cyclic sand behaviour. Special mention in this context 42 goes to the SANISAND04 model proposed by Dafalias and Manzari (2004), built on Manzari and 43 Dafalias (1997) and forefather of several later formulations (Zhang and Wang 2012; Boulanger and 44 Ziotopoulou 2013; Dafalias and Taiebat 2016; Petalas et al. 2019). Among these, the PM4Sand 45 model (Boulanger and Ziotopoulou 2013; Ziotopoulou and Boulanger 2016) possesses remarkable 46 capabilities to reproduce undrained cyclic behaviour, including the simulation of pore pressure 47 build-up, liquefaction triggering and, in medium-dense/dense sands, 'cyclic mobility' (Elgamal 48

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et al. 2003) – in turn associated with transient regains in shear resistance, and gradual shear
strain accumulation at vanishing confinement. Cyclic mobility is relevant to the serviceability of
earth structures and foundations under prolonged cyclic loading (Ziotopoulou and Boulanger 2016;
Kementzetzidis et al. 2019), as well as to seismic site response (Roten et al. 2013).

Recently, Liu et al. (2019) enhanced the SANISAND04 formulation by introducing the concept 53 of memory surface (MS) (Stallebrass and Taylor 1997; Maleki et al. 2009; Corti et al. 2016) to 54 better account for fabric-related effects and their impact on cyclic ratcheting behaviour (Houlsby 55 et al. 2017). The model – henceforth referred to as SANISAND-MS – can predict variations in soil 56 stiffness and strain accumulation under thousands of drained loading cycles (high-cyclic loading). 57 The same modelling features also allow better simulation of the undrained hydro-mechanical 58 response, especially in terms of extent and timing of cyclic pore pressure accumulation (Liu et al. 59 2018). It was noted, however, that further improvements would be needed to unify the simulation 60 of undrained cyclic behaviour over a wide range of initial sand densities and loading conditions 61 (Liu et al. 2018). 62

This work takes further the success of SANISAND-MS as presented in Liu et al. (2019), with reference to undrained cyclic loading. Besides the ability of capturing liquefaction triggering, the emphasis of this work lies on the following aspects: (i) cyclic pore pressure build-up, including its cycle-by-cycle timing in the pre-liquefaction stage; (ii) stress-strain response in the post-liquefaction phase (cyclic mobility behaviour); and (iii) influence of previous loading history on the undrained cyclic response. These objectives are accomplished without compromising the previous achievements of Liu et al. (2019).

The performance of the upgraded SANISAND-MS formulation is inspected in detail, and thoroughly validated against the experimental datasets from Wichtmann (2005) and Wichtmann and Triantafyllidis (2016) – including undrained cyclic triaxial tests on both isotropically and anisotropically consolidated sand specimens. The present research is largely motivated by current offshore wind developments, where the need for advanced analysis of cyclic soil-foundation interaction is particularly felt (Pisanò 2019).

#### 76 UPGRADED SANISAND-MS FORMULATION

#### 77 Notation

Stresses are meant as 'effective' throughout the paper, bold-face notation is used for tensor 78 quantities, and the symbol ':' stands for inner tensor product. Stresses and strains are represented 79 by the tensors  $\boldsymbol{\sigma}$  and  $\boldsymbol{\varepsilon}$ , with typical tensor decompositions including: deviatoric stress  $\boldsymbol{s} = \boldsymbol{\sigma} - p\boldsymbol{I}$ , 80 with  $p = \text{tr}\boldsymbol{\sigma}/3$  effective mean stress and  $\boldsymbol{I}$  identity tensor; deviatoric strain  $\boldsymbol{e} = \boldsymbol{\varepsilon} - (\varepsilon_{vol}/3)\boldsymbol{I}$ , with 81  $\varepsilon_{vol} = tr \boldsymbol{\varepsilon}$  volumetric strain – superscripts e and p are used to denote 'elastic' and 'plastic' strain 82 components. The deviatoric stress ratio tensor is defined as  $\mathbf{r} = \mathbf{s}/p$ . The deviatoric stress q is 83 defined as  $q = \sqrt{3J_2}$ , with  $J_2$  second invariant of **s**. The symbols 'tr' and ' $\langle \rangle$ ' indicate trace and 84 Macauley brackets operators, respectively. 85

#### 86 Background

The proposed version of SANISAND-MS upgrades the formulation by Liu et al. (2019), built 87 on the SANISAND04 bounding surface model (Dafalias and Manzari 2004) and enriched with 88 the notion of memory surface (Corti et al. 2016), which replaces the fabric tensor of the original 89 formulation. The general representation of all model loci in the normalised deviatoric stress ratio 90 plane is provided in Fig.1. The model formulation is founded on the critical state theory and 91 makes use of: (1) a narrow conical yield locus (f) enclosing the elastic domain; (2) a wide conical 92 bounding surface  $(f^B)$ , setting stress bounds compliant with an evolving state parameter  $\Psi$  (Been 93 and Jefferies 1985) as per Manzari and Dafalias (1997); (3) a conical dilatancy surface  $(f^D)$ , 94 separating stress zones associated with contractive and dilative deformations as a function of  $\Psi$ 95 (Manzari and Dafalias 1997; Li and Dafalias 2000; Dafalias and Manzari 2004); (4) a conical 96 memory surface  $(f^M)$ , bounding an evolving stress region related to increased hardening response 97 due to 'non-virgin' loading and, in turn, stress-induced anisotropy at the micro-scale. The memory 98 surface enables phenomenological representation of fabric changes induced by the cyclic loading 99 history, such as variations in stiffness and dilatancy. The memory mechanism takes place in the 100 multi-dimensional stress space and is intrinsically sensitive to the loading direction. 101

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The model features non-associated plastic flow and, owing to the state parameter mechanism,

<sup>103</sup> is able to reproduce sand behaviour over a wide range of void ratios via a single set of parameters. <sup>104</sup> Several modelling ingredients – e.g., elastic relationships, deviatoric plastic flow, critical state line <sup>105</sup> (CSL) and model surfaces – are directly inherited from Liu et al. (2019). The use of the yield <sup>106</sup> back-stress ratio  $\boldsymbol{\alpha}$  is resumed here as in Dafalias and Manzari (2004) to avoid certain numerical <sup>107</sup> inconveniences, so that its projections onto bounding, dilatancy and critical state surfaces are <sup>108</sup> employed in the model formulation. For brevity, already published constitutive equations are only <sup>109</sup> reported in Appendix A, while main focus is on defining and validating new model features.

#### 110 New features

New relationships for memory surface evolution, plastic flow rules and hardening laws are presented in this section and summarised in Appendix A. The new model ingredients do not affect the capabilities of the previous formulation, but do influence the calibration of certain cyclic parameters inherited from Liu et al. (2019). Calibration and role of newly defined parameters are discussed in what follows. Ideally, four extra-tests would be needed for their calibration, including stress-controlled undrained cyclic triaxial tests at different relative densities and cyclic stress ratios. Nevertheless, the upgraded model can be reduced to a 'lighter' version whenever convenient.

The implications of the mentioned improvements are elucidated by comparing previous and latest SANISAND-MS simulations of triaxial test results from Wichtmann and Triantafyllidis (2016). The reference cyclic undrained tests were performed on Karlsruhe fine sand ( $D_{50} =$ 0.14mm,  $C_u = D_{60}/D_{10} = 1.5$ ,  $e_{max} = 1.054$ ,  $e_{min} = 0.677$ ). Simulations of the previous SANISAND-MS model (Liu et al. 2019) are related to the soil parameters given in Appendix A from Liu et al. (2018).

#### 124 Memory surface and its evolution

The memory surface  $(f^M)$  tracks stress states already experienced by the sand during its (cyclic) loading history. It accounts for fabric changes and load-induced anisotropy via the evolution of its size  $(m^M)$  and back-stress ratio  $(\boldsymbol{\alpha}^M)$  (Corti et al. 2016; Liu et al. 2019; Liu and Pisanò 2019). The expansion of the memory surface (i.e., increase in  $m^M$ ) corresponds to the experimental observation of sand becoming stiffer as fabric is reinforced by cycling within the 'non-virgin' domain. On the other hand, the occurrence of dilation causes loss of sand stiffness (Nemat-Nasser and Tobita 1982), which can be reproduced by the model through a decrease in  $m^M$ . This experimental evidence led to postulate a parallel shrinking mechanism for the memory surface, so that the change in memory surface size  $(dm^M)$  is decomposed into two terms: a memory surface expansion term  $dm_+^M$  and a memory surface contraction term  $dm_-^M$ :

$$dm^M = dm^M_+ + dm^M_- \tag{1}$$

Enforcing plastic consistency under 'virgin loading' (i.e., with tangent yield and memory surfaces at the current stress point  $\sigma$  and the memory surface has no influence on soil stiffness, see Liu et al. (2019)) in the contractive regime allows to derive the (positive) expansion rate  $dm_{+}^{M}$ :

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$$dm_{+}^{M} = \sqrt{3/2} d\boldsymbol{\alpha}^{M} : \boldsymbol{n}$$
<sup>(2)</sup>

where *n* is the unit tensor normal to the yield surface *f* (Fig.2a). As discussed in Liu et al. (2019), variations in size and location of the memory surface are inter-related.  $d\boldsymbol{\alpha}^{M}$  describes the translation of the memory surface centre, assumed to take place along the direction of  $\boldsymbol{\alpha}^{b} - \boldsymbol{r}_{\alpha}^{M}$ :

$$d\boldsymbol{a}^{M} = 2/3 \langle L \rangle h^{M} (\boldsymbol{a}^{b} - \boldsymbol{r}_{\alpha}^{M})$$
(3)

in which  $\boldsymbol{\alpha}^{b}$  is the bounding back-stress ratio (Fig.2a) and  $\boldsymbol{r}_{\alpha}^{M} = \boldsymbol{\alpha}^{M} + \sqrt{2/3}(m^{M} - m)\boldsymbol{n}$  (different from the memory image point  $\boldsymbol{r}^{M} = \boldsymbol{\alpha}^{M} + \sqrt{2/3}m^{M}\boldsymbol{n}$  in Fig.1). *L* is the plastic multiplier (Appendix A), while  $h^{M}$  is the counterpart of the hardening coefficient defined with respect to the memory surface — its expression is specified later on.

As a new feature, the shrinkage rate of the memory surface  $dm_{-}^{M}$  is further linked to the induced cumulative expansion of the memory surface size  $m_{+}^{M} = \int dm_{+}^{M}$  over the whole loading history experienced from a known initial state. The introduction of the term  $m_{+}^{M}$ , monotonically increasing under shearing and consequent plastic straining, ensures rapid degradation of the memory surface at large strain levels. Therefore, virgin loading conditions are quickly reinstated upon load increment
 reversal after severe dilation (due to inhibited memory surface effects). This feature is consistent
 with the observations of Yimsiri and Soga (2010) and Ziotopoulou and Boulanger (2016), who
 noted that sand behaviour at large strain levels is mainly governed by the current relative density:

$$dm_{-}^{M} = -\frac{m^{M}}{\zeta} f_{shr} \langle b_{r}^{b} \rangle m_{+}^{M} \left\langle -d\varepsilon_{vol}^{p} \right\rangle$$
(4)

where  $\zeta$  is a parameter governing the shrinking rate of the memory surface, while the geometrical factor  $f_{shr}$  ensures that the memory surface never becomes smaller than the elastic domain (see Appendix 1 in Liu et al. (2019) for details):

$$f_{shr} = 1 - (x_1 + x_2)/x_3 \tag{5}$$

with  $x_{1,2,3}$  illustrated in Fig.2b and defined as:

$$x_{1} = \boldsymbol{n}^{M} : (\boldsymbol{r}^{M} - \boldsymbol{r})$$

$$x_{2} = \boldsymbol{n}^{M} : (\boldsymbol{r} - \tilde{\boldsymbol{r}})$$

$$x_{3} = \boldsymbol{n}^{M} : (\boldsymbol{r}^{M} - \tilde{\boldsymbol{r}}^{M})$$
(6)

<sup>163</sup> In Eq.6:

$$\tilde{\boldsymbol{r}} = \boldsymbol{\alpha} - \sqrt{2/3}m\boldsymbol{n}$$
  $\tilde{\boldsymbol{r}}^M = \boldsymbol{\alpha}^M - \sqrt{2/3}m^M\boldsymbol{n}$  (7)

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and  $\mathbf{n}^{M}$  is the unit tensor oriented parallel to  $(\mathbf{r}^{M} - \mathbf{r})$  (see Fig.2b):

$$\boldsymbol{n}^{M} = (\boldsymbol{r}^{M} - \boldsymbol{r}) / \sqrt{(\boldsymbol{r}^{M} - \boldsymbol{r}) : (\boldsymbol{r}^{M} - \boldsymbol{r})}$$
(8)

The term  $\langle b_r^b \rangle$  in Eq.4 is also introduced to properly handle strain-softening stages: during strain softening,  $(\boldsymbol{\alpha}^b - \boldsymbol{\alpha}) : \boldsymbol{n} < 0$ , which may results in  $b_r^b = (\boldsymbol{\alpha}^b - \boldsymbol{r}_{\alpha}^M) : \boldsymbol{n} < 0$  and contemporary shrinkage of both bounding and memory surfaces may occur. As a consequence,  $dm_+^M < 0$  and  $m_+^M$  may decrease, which would be in contrast with the assumption of non-decreasing  $m_+^M$ . The following expression of the memory surface hardening coefficient  $h^M$  in Eqs. 2–3 results from derivations similar to those in Liu et al. (2019) (see Table 1):

$$h^{M} = \frac{1}{2} \left( \tilde{h} + \hat{h} \right) = \frac{1}{2} \left[ \frac{b_{0}}{(\boldsymbol{r}_{\alpha}^{M} - \boldsymbol{\alpha}_{in}) : \boldsymbol{n}} + \sqrt{\frac{3}{2}} \frac{m^{M} m_{+}^{M} f_{shr} \langle b_{r}^{b} \rangle \langle -D \rangle}{\zeta(\boldsymbol{\alpha}^{b} - \boldsymbol{r}_{\alpha}^{M}) : \boldsymbol{n}} \right]$$
(9)

where  $b_0$  is the hardening factor given by Dafalias and Manzari (2004) (Appendix A), and  $\boldsymbol{\alpha}_{in}$ the back-stress ratio at stress increment reversal. Closer inspection of Eq. 9 leads to recognise the chance of a vanishing denominator in  $\hat{h}$  (e.g., if either  $\boldsymbol{\alpha}^b = \boldsymbol{r}^M_{\alpha}$  or  $\boldsymbol{n} \perp (\boldsymbol{\alpha}^b - \boldsymbol{r}^M_{\alpha})$ ), which may abruptly accelerate the evolution of  $\boldsymbol{\alpha}^M$  and temporarily leave the yield locus outside the (shrinking) memory surface. The effects of such occurrence, rare but possible, may be mitigated in the numerical implementation of the model, for instance by inhibiting shrinkage of the memory surface when becoming tangent to the yield surface.

Overall, the above upgraded laws for memory surface evolution allow to erase fabric effects at large strain levels, in agreement with available experimental evidence (Yimsiri and Soga 2010; Ziotopoulou and Boulanger 2016).

184 Dilatancy

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The model proposed by Liu et al. (2019) can already predict liquefaction triggering (according 185 to Seed and Lee (1966), the first occurrence of  $p' \approx 0$ ), and provides for medium-dense/dense 186 sands reasonable stress path shapes in the post-dilation phase ('butterfly-shaped' q - p response). 187 However, accurate simulation of peculiar stress-strain loops during cyclic mobility is beyond the 188 possibilities of that model. Ammending this short-coming requires introducing changes to the 189 formulation governing sand dilatancy. Indeed, as discussed by Elgamal et al. (2003) and Boulanger 190 and Ziotopoulou (2013), the modelling of cyclic mobility is intimately related to the description of 191 sand dilatancy. Within the SANISAND framework, the dilatancy coefficient D in the plastic flow 192 rule is generally expressed as (Appendix A): 193

 $D = A_d d \tag{10}$ 

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$$d = (\boldsymbol{\alpha}^d - \boldsymbol{\alpha}) : \boldsymbol{n} \tag{11}$$

and  $\boldsymbol{\alpha}^{d}$  represents the image back-stress ratio on the dilatancy surface. In Liu et al. (2019), the term  $A_{d}$  was already set to depend on the sign of plastic volume changes (i.e., contraction or dilation) before the previous load increment reversal through the term  $\langle \tilde{b}_{d}^{M} \rangle = \langle (\tilde{\boldsymbol{\alpha}}^{d} - \tilde{\boldsymbol{r}}_{\alpha}^{M}) : \boldsymbol{n} \rangle$ . Such a dependence was introduced to capture the increase in pressure build-up upon post-dilation load increment reversals — a phenomenon that Dafalias and Manzari (2004) reproduced through the concept of fabric tensor. Compared to Liu et al. (2019), the definition of  $A_{d}$  is here enhanced with some new features, mainly instrumental to the simulation of undrained cyclic mobility:

- in case of (plastic) contraction ( $d \ge 0$ ) following previous contraction ( $\tilde{b}_d^M \le 0$ ):

$$A_d = A_0 \tag{12}$$

- in case of (plastic) contraction ( $d \ge 0$ ) following previous dilation ( $\tilde{b}_d^M > 0$ )

 $A_d = A_0 \exp\left[\beta_1 F\left(\frac{p}{p_{max}}\right)^{0.5}\right] g^k(\theta)$ (13)

- in case of dilation (d < 0)

$$A_d = A_0 \exp\left[\beta_2 F\left(1 - \left(\frac{p}{p_{max}}\right)^{0.5}\right) \frac{d}{||\boldsymbol{a}^c||}\right] \frac{1}{g(\theta)}$$
(14)

In the above relationships,  $A_0$  is the 'intrinsic' dilatancy parameter already present in Dafalias and Manzari (2004).  $||\boldsymbol{\alpha}^c||$  in Eq.14 is the Euclidean norm of  $\boldsymbol{\alpha}^c$  (see Appendix A) introduced for normalisation purposes, which represents the distance between the origin of the deviatoric stress ratio plane and the image back-stress ratio on the critical surface  $f^C$  (Fig.1). The new dilatancy features in Eqs.13-14 are phenomenologically associated with the following mechanical factors:

• **Fabric history** 

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F is a non-decreasing scalar variable related to the previous history of fabric evolution:

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$$F = \ln\left[1 + \frac{|m_{-}^{M}|}{(|m_{+}^{M}| + |m_{-}^{M}|)^{0.5}}\right] = \ln\left[1 + \frac{\int |dm_{-}^{M}|}{(\int |dm_{+}^{M}| + \int |dm_{-}^{M}|)^{0.5}}\right]$$
(15)

F plays a similar role as the 'damage index' in Boulanger and Ziotopoulou (2013), that is 218 to progressively degrade  $A_d$  at increasing number of cycles. This feature helps reproducing 219 progressive shear strain accumulation, for instance in undrained DSS tests with imposed 220 symmetric shear loading (Arulmoli et al. 1992; Andersen 2009). The effect of this mod-221 elling ingredient can be appreciated by comparing model simulations in Fig.3a and Fig.3b, 222 performed with previous and upgraded SANISAND-MS, respectively. It should also be 223 noted that, as F is a non-decreasing variable, it will permanently have an influence also 224 on the post-cyclic response, possibly featuring different drainage conditions. Post-cyclic 225 drained behaviour, for instance, would be more (less) contractive (dilative) than without the 226 use of F in the flow rule. There is hardly any experimental evidence available to either sup-227 port or falsify such occurrence, so that caution is recommended when applying the model 228 to problems with very variable drainage conditions and/or distinct stages of consolidation. 229

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#### Sensitiveness to stress state and path

Dependence on the (relative) Lode angle function  $(g(\theta))$  and the term  $d/||\boldsymbol{\alpha}^c||$  were sug-231 gested by experimental results as a way to modulate the response, and particularly strain 232 accumulation, with respect to different cyclic stress paths (e.g., triaxial or simple shear). 233 Typical simulation results of previous and upgraded SANISAND-MS models are shown in 234 Figs.4a and 4b, respectively. The pressure term  $(p/p_{max})^{0.5}$   $(p_{max}$  is the highest effective 235 mean pressure ever experienced) reflects the higher proneness to shear straining observed 236 at very low effective stress levels, progressively reducing at increasing p – see Fig.3b and 237 Fig.4b. 238

<sup>239</sup> Dilatancy features in the upgraded model can be tuned to experimental data through the material <sup>240</sup> parameters  $\beta_1$  and  $\beta_2$  in Eqs.13 and 14. These parameters govern cyclic shear straining in the

dilative regime – cyclic volume changes before any dilation mostly depend on the parameter  $A_0$ 241 and the memory-hardening parameter  $\mu_0$  in Appendix A. Sound calibration of  $\beta_1$  requires data 242 from undrained cyclic triaxial tests in which initial liquefaction is triggered. As exemplified in 243 Fig.5, the parameter  $\beta_1$  influences the undrained triaxial stress-strain response in terms of ultimate 244 normalised accumulated pore pressure (throughout this work, pore water pressure generation is 245 tracked at the end of each full cycle when  $q = q_{ave}$  level). Larger  $\beta_1$  results in higher  $u^{acc}/p_{in}$ 246 ratios (i.e., smaller residual effective stress). For the considered Karlsruhe fine sand  $\beta_1 = 4$  was 247 selected, with  $\beta_2$  negligibly affecting the final  $u^{acc}$  level. 248

At given  $\beta_1$ , increasing  $\beta_2$  results in larger accumulation of cyclic shear strain in undrained 249 cyclic DSS tests (see Fig.3b). Unfortunately, in the lack of undrained cyclic DSS tests performed 250 on the same Karlsruhe sand,  $\beta_2$  had to be identified, together with k in Eq.13, by a trial-and-251 error procedure. In the case of triaxial loading, increasing  $\beta_2$  determines larger cyclic axial strain 252 (see Fig.6b), whereas the parameter k in Eq.13 governs the influence of the stress path through 253 the relative Lode angle  $\theta$  in Fig.1. Fig.6b shows that, for a cyclic triaxial test, higher k results in 254 positive/compressive cyclic axial strains larger than on the negative/extension side. The comparison 255 to Wichtmann and Triantafyllidis (2016)'s triaxial test results (Fig.6a) led to identify the parameter 256 pair  $\beta_2 = 3.2$  and k = 2. Two remarks about formulation and limitations of the new flow rule: 257

- 1. The piece-wise definition of  $A_d$  implies discontinuity in the dilatancy coefficient D when the material transits from contractive to dilative behaviour (i.e., when the yield locus crosses the dilatancy surface) – even in presence of continuous variations in stress ratio r (thus, in loading direction n). Consequently, continuity of volumetric plastic strain increments may not be guaranteed, similarly to Boulanger and Ziotopoulou (2013) and Khosravifar et al. (2018);
- In contrast with the (inconclusive) findings of some experimental studies, the model predicts
   unlimited strain accumulation during cyclic mobility compare to Fig.6a, where only
   limited strain increments are observed in the last few loading cycles. While other modelling
   assumptions are certainly possible (Barrero et al. 2019), the latter point will receive further

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attention when broader consensus about underlying physical mechanisms is reached (Wang and Wei 2016; Wang et al. 2016).

#### 270 Hardening coefficient

In its first version, SANISAND-MS had limited capability to quantitatively reproduce com-271 plex relationships between cyclic pore pressure accumulation and relevant loading factors. Fig.7 272 compares the performance of previous SANISAND-MS (blue lines) (Liu et al. 2019) in repro-273 ducing Wichtmann and Triantafyllidis (2016)'s triaxial data (blacks lines) regarding undrained 274 pre-liquefaction behaviour under cyclic symmetric loading at varying cyclic amplitude ratios 275  $(\eta_{ampl} = q_{ampl}/p_{in})$ , with  $q_{ampl}$  the cyclic shear amplitude and  $p_{in}$  the initial mean effective 276 stress). The previous SANISAND-MS predicts more limited variation in the number of loading 277 cycles  $N_{ini}$  to trigger initial liquefaction  $(u^{acc}/p_{in} \approx 1 \text{ for the first time})$ . 278

The comprehensive database of Wichtmann and Triantafyllidis (2016) supports the idea that more cycles are required to trigger liquefaction (higher  $N_{ini}$ ) at low  $\eta_{ampl}$ . It could thus be attempted to link the increase in  $N_{ini}$  to higher values of the hardening coefficient *h* through explicit dependence on  $\eta_{ampl}$ . However, as  $\eta_{ampl}$  cannot be a priory defined in general boundary value problems, the current stress ratio  $\eta$  instead of  $\eta_{ampl}$  is adopted in the upgraded definition of the hardening coefficient *h*:

$$h = \frac{b_0}{(\boldsymbol{\alpha} - \boldsymbol{\alpha}_{in}) : \boldsymbol{n}} \exp\left[\mu_0 \left(\frac{p}{p_{atm}}\right)^{0.5} \left(\frac{b^M}{b_{ref}}\right)^{w_1} \frac{1}{\eta^{w_2}}\right]$$
(16)

where  $\eta = q/p = \sqrt{3J_2}/p$  (see *Notation* section).  $b^M$  represents the distance between the current back-stress ratio  $\boldsymbol{\alpha}$  and its image point  $\boldsymbol{r}_{\alpha}^M$  on the memory surface, while  $b_{ref}$  is a reference normalisation factor (Appendix A). The term  $1/\eta^{w_2}$  (with  $w_2$  new model parameter), explicitly accounts for the deviatoric span of the loading path – for more robust numerical implementation, m (radius of the yield surface in the stress ratio  $\pi$  plane) is set as  $\eta$ 's lower bound.

Input to the calibration of the  $w_2$  parameter can be obtained from the experimental relationship between  $N_{ini}$  and  $\eta_{ampl}$  in triaxial tests on isotropically consolidated sand. As mentioned above,

increase in  $N_{ini}$  is linked to higher values of the hardening coefficient h, which is in turn inversely 293 related to  $\eta_{ampl}$  (i.e.,  $N_{ini} \propto h \propto [\exp(\text{factor} \cdot 1/\eta_{ampl}^{w_2})])$ . Such observation prompted the investiga-294 tion of the relationship between  $\ln(N_{ini})$  and  $1/\eta_{ampl}^{w_2}$ ). It was concluded that for fixed  $\eta_{ampl}$ , dense 295 sands (i.e., with  $D_{r0}$  larger than critical) experience more loading cycles before liquefaction. In 296 summary, the experimental relationship between  $\ln(N_{ini})/D_{r0}$  and  $1/(\eta_{ampl}^{w_2})$  emerging from a set 297 of tests is proposed as a tool to calibrate  $w_2$  – see Fig.8. This requires at least four stress-controlled 298 undrained triaxial tests on isotropically consolidated specimens, at varying  $\eta_{ampl}$  and  $D_{r0}$ , until 299 cyclic liquefaction is triggered. However, since in Eq.16 the current stress ratio  $\eta$  is adopted instead 300 of directly using  $\eta_{ampl}$ , the calibrated  $w_2$  may need further adjustment together with  $w_1$  and  $\mu_0$  (for 301 which calibration procedures are given in the following section). Should available data be insuffi-302 cient,  $w_2 = 0$  is suggested as an initial value, and followed with a sensitivity study to determine its 303 relevance and possibly motivate the gathering of the data for its calibration. 304

The other exponent  $w_1$  in Eq.16 was pre-set to 2 in Liu et al. (2019) for simplicity. Herein,  $w_1$  is re-activated as a free model parameter for more flexibility. Its value, together with  $\mu_0$ 's, was calibrated mostly by trial-and-error, starting from the default setting  $w_1 = 2$ . The same test data-set used for calibrating  $w_2$  can also support the identification of  $w_1$  when looking at pore pressure accumulation trends, e.g., in terms of  $u^{acc}/p_{in}$  versus number of loading cycles. Fig.9 shows that good agreement for the examined Karlsruhe sand is achieved for  $\mu_0 = 65$  and  $w_1 = 2.5$ .

Fig.7 also shows the performance of upgraded SANISAND-MS (red lines). As discussed in the following section, the upgraded model appears better suited to capture the dependence of  $N_{ini}$ (number of cycles to liquefaction) on the cyclic stress amplitude at different relative densities.

314

#### PREDICTION OF UNDRAINED CYCLIC RESPONSE

This section demonstrates the predictive capabilities of the model with respect to undrained cyclic loading. Using the set of calibrated parameters in Table 3, the model performance is assessed against additional triaxial test results on Karlsruhe fine sand (Wichtmann and Triantafyllidis 2016), not previously used for calibration.

#### **Response of isotropically consolidated sand**

#### 320 Cyclic pore pressure accumulation

Cyclic build-up of pore pressure may cause stiffness and strength losses (cyclic liquefaction), 321 for instance during seismic events. Many empirical models have been developed (Dobry et al. 322 1985; Idriss and Boulanger 2006; Ivšić 2006; Chiaradonna et al. 2018) to simplify the prediction 323 of such build-up by directly relating the pore pressure ratio  $(u^{acc}/p_{in})$  to the ratio between current 324 number of cycles (N) and total number of cycles to liquefaction  $(N_{ini})$ . It seems interesting to 325 verify how pore pressure predictions from SANISAND-MS (both previous and upgraded versions) 326 compare to empirical models, such as that recently proposed by Chiaradonna et al. (2018). In 327 Fig.10, SANISAND-MS and empirical model predictions are compared to experimental data from 328 Wichtmann and Triantafyllidis (2016), concerning triaxial tests performed at varying cyclic stress 329 amplitude ratio. Although both plasticity and empirical models reproduce well experimental data, 330 it is worth noting that the simulation of pore pressure accumulation trends is usually easier when 331 pursued in terms of normalised number of cycles  $N/N_{ini}$ . It is shown hereafter that reproducing 332 the absolute  $N_{ini}$  value poses a more serious challenge for constitutive modelling. 333

Influence of initial effective mean pressure Experimental test results from Wichtmann and 334 Triantafyllidis (2016) (Fig.11) show that it is not straightforward to interpret the influence of 335 the initial consolidation pressure  $p_{in}$  in tests featuring constant cyclic stress amplitude ratio 336  $(\eta_{ampl} = q_{ampl}/p_{in})$ . Axial strain accumulation in the cyclic mobility stage does not show obvious 337 dependence on  $p_{in}$  either. Simulation results obtained with the upgraded SANISAND-MS formu-338 lation support similar conclusions (Fig.11b). For instance, the considered cases with  $\eta_{ampl} = 0.25$ 339 and  $p_{in} = 100, 200, 300$  kPa are associated in experiments with  $N_{ini}$  values equal to 100, 77 and 110, 340 respectively – i.e., with no monotonic dependence of  $N_{ini}$  on  $p_{in}$  (and arguably with an influence 341 of specimen preparation). Overall, the proposed SANISAND-MS formulation shows good ability 342 to predict the impact of  $p_{in}$  both in terms of pore pressure build-up and strain accumulation with 343 the upgraded formulation performing better than its previous version. 344

**Influence of cyclic amplitude ratio** The reference experimental data show that higher values of 345 the cyclic amplitude stress ratio ( $\eta_{ampl} = q_{ampl}/p_{in}$ ) result in faster triggering of liquefaction (i.e., 346 lower  $N_{ini}$ ) – see Fig.12a and Fig.12e. Both SANISAND-MS versions prove sensitive to this effect 347 (see Fig.12b and Fig.12e). However, while Liu et al. (2019)'s formulation largely underestimates 348  $N_{ini}$  for  $\eta_{ampl} = 0.2$  and 0.25, the upgraded model predicts accurate  $N_{ini}$  values in all considered 349 cases. This confirms the effectiveness of the new hardening modulus definition in Eq.16. Further, 350 the upgraded formulation captures well the axial strain accumulation, both on positive and negative 351 sides (compare Fig.12c and Fig.12d). 352

**Influence of initial relative density** Wichtmann and Triantafyllidis (2016)'s data also confirm 353 the expectation that, under given conditions, the effective mean pressure vanishes faster at lower 354 initial relative density (see stress paths in Fig.13a and Fig.13e). Both SANISAND-MS versions 355 succeed also in this respect (Fig.13b and Fig.13e). Nonetheless, the new formulation improves 356 quantitative pore pressure predictions owing to the new material parameter  $w_2$ , which scales cyclic 357 amplitude effects with respect to the void ratio (see Eq.16 and Fig.9) - compare experimental data 358 and upgraded model predictions in Figs. 13a to 13b). The new model, however, seems to reproduce 359 the influence on strain accumulation of the initial relative density (Figs.13c to 13d) less accurately 360 than of other input factors (Figs.11 - 12). 361

# 362

#### **Response of anisotropically consolidated sand**

SANISAND-MS was further challenged to reproduce the undrained response of anisotropically 363 consolidated sand specimens. Useful insight in this respect can be obtained from the comparison 364 in Fig.14 between effective stress paths from experimental results (Wichtmann and Triantafyllidis 365 2016) and SANISAND-MS simulations. In particular, cases with cyclic stress amplitude ratio 366  $(\eta_{ampl} = q_{ampl}/p_{in})$  smaller or larger than the initial average stress ratio  $(\eta_{ave} = q_{ave}/p_{in})$  were 367 considered in both experiments and simulations - Figs.14a, 14b. Fig.14 suggests that, when 368  $\eta_{ampl} < \eta_{ave}$  (i.e., with no compression-to-extension reversals in terms of current cyclic stress 369 ratio, Fig.14a), effective stress paths evolve towards steady loops after a few loading cycles – with 370

<sup>371</sup> no liquefaction triggering ( $u^{acc}/p_{in} < 1$ ). This occurrence corresponds with the attainment of a <sup>372</sup> pore pressure plateau in  $u^{acc}/p_{in} - N$  plots (Fig.14c). Further, the characteristic butterfly shape <sup>373</sup> of the steady stress path is well captured for  $\eta_{ampl} > \eta_{ave}$  (see Fig.14b). When compared to <sup>374</sup> laboratory data, SANISAND-MS simulations reproduce quite well such experimental evidence, <sup>375</sup> including reasonable timing of effective mean pressure reduction against the number of cycles <sup>376</sup> (Fig.14c), especially for  $\eta_{ampl} > \eta_{ave}$ .

377

### Influence of drained cyclic pre-loading

It is well-known that previous loading history affects the hydro-mechanical response of sands 378 to undrained cyclic loading, including their susceptibility to liquefaction. In this section the impact 379 of drained cyclic pre-loading on subsequent undrained pore pressure build-up is explored. To this 380 end, results from a different experimental database were considered. Fig.15 shows SANISAND-MS 381 simulation results for the quartz sand tested by Wichtmann (2005) ( $D_{50} = 0.55$  mm,  $D_{10} = 0.29$ 382 mm,  $C_u = D_{60}/D_{10} = 1.8$ ,  $e_{max} = 0.874$ ,  $e_{min} = 0.577$ ), corresponding with  $p_{in} = 100$  kPa, 383  $e_{in} = 0.684$ , undrained cyclic stress amplitude  $q_{ampl}^{pre} = 45$  kPa. The model parameters calibrated 384 for this second sand are reported in Table 3. Monotonic parameters and  $\mu_0$  (i.e., from  $G_0$  to  $\mu_0$ 385 in Table 3) coincide with those calibrated by Liu et al. (2018) and Liu et al. (2019), while the 386 aforementioned default values  $w_1 = 2$  and  $w_2 = 0$  were assumed;  $\beta_1$ ,  $\beta_2$ , k and  $\zeta$  were calibrated 387 against the deviatoric stress-axial strain response from only one stress-controlled triaxial test at 388 constant cyclic amplitude. 389

<sup>390</sup> Upgraded SANISAND-MS simulations were carried out for three different cases: (1) without <sup>391</sup> drained pre-loading cycles; (2) with 10 drained pre-cycles of amplitude  $q_{ampl}^{pre} = 30$  kPa, followed <sup>392</sup> by undrained cyclic loading; (3) with 10 drained pre-cycles of amplitude  $q_{ampl}^{pre} = 50$  kPa, followed <sup>393</sup> by undrained cyclic loading. It is generally observed that drained cyclic pre-loading under the <sup>394</sup> phase-transformation line tends to delay the onset of liquefaction (i.e., to increase  $N_{ini}$ , see q - p<sup>395</sup> stress paths in Figs.15a–15c).

<sup>396</sup> Simulation results in Fig.15d (red lines) are in very good agreement with experimental measure-<sup>397</sup> ments (black lines) in terms of pore water pressure accumulation, and support the suitability of the

adopted memory surface framework. In essence, applying drained cyclic pre-loading contributes 398 to the "reinforcement" of sand fabric. This aspect is phenomenologically tracked by the model 399 through the corresponding evolution of the memory surface size/location, and thus exploited to 400 re-tune soil stiffness and dilatancy. The larger  $m^M$ , the higher the resistance to liquefaction, i.e., 401 the larger  $N_{ini}$ . As highlighted in Fig. 15e, accurate simulation of effective stress paths enables to 402 reliably predict the dependence of  $N_{ini}$  on the amplitude of drained pre-cycles. It is finally worth 403 noting that the parent SANISAND04 model (Dafalias and Manzari 2004) would be practically 404 insensitive to drained cyclic pre-loading, except for the effect of a slightly different void ratio at the 405 beginning of undrained cycling. 406

#### 407

#### **CONCLUDING REMARKS**

The memory-enhanced bounding surface model proposed by Liu et al. (2019), SANISAND-408 MS, was improved to reproduce essential features of the hydro-mechanical response of sands to 409 undrained cyclic loading. The previous mathematical formulation was upgraded by: (i) modifying 410 memory surface evolution laws to better reflect fabric effects at larger strains; (ii) enhancing the 411 description of sand dilatancy through new terms accounting for fabric evolution history, and stress 412 state/path; (iii) incorporating a deviatoric stress ratio term into the hardening modulus. While ready 413 application to 3D boundary value problems was the main motivation of such effort, a few aspects 414 of the proposed constitutive model will require further research in the near future, for instance 415 to: (a) avoid discontinuities in the dilatancy formulation; (b) more flexibly model deviatoric strain 416 accumulation during cyclic mobility, e.g., by allowing for strain saturation limits if observed in 417 experimental data; (c) investigate the evolution of fabric history effects through varying drainage 418 conditions. 419

The above modifications enabled substantial improvement of simulated pore pressure build-420 up and cyclic mobility, with sound sensitiveness to the main governing factors. After parameter 421 calibration, the model was thoroughly validated against published results of undrained cyclic 422 triaxial tests. Further qualitative insight into the expected effect of different loading conditions 423 (e.g., under simple shear loading). The upgraded SANISAND-MS model confirmed the suitability 424

- of combining the memory surface concept with the well-established bounding surface plasticity
- 426 framework.

## 427 APPENDIX A: UPGRADED SANISAND-MS CONSTITUTIVE EQUATIONS

428

FEATURE	EQUATION	PARAMETER
Elasticity	$G = G_0 p_{atm} (2.97 - e)^2 / (1 + e) \sqrt{p/p_{atm}}$ K = 2(1 + v)G / [3(1 - 2v)]	$G_0$ dimensionless shear modulus y Poisson ratio
Critical	(-)	$e_0$ reference critical void ratio
state line	$e_c = e_0 - \lambda_c (p_c / p_{atm})^{\circ}$	$\lambda_c, \xi$ CSL shape parameters
Yield surface	$f = \sqrt{(\boldsymbol{s} - p\boldsymbol{\alpha}) : (\boldsymbol{s} - p\boldsymbol{\alpha})} - \sqrt{2/3}pm$	<i>m</i> yield locus opening parameter
Memory surface	$f^M = \sqrt{(\boldsymbol{s} - p\boldsymbol{\alpha}^M) : (\boldsymbol{s} - p\boldsymbol{\alpha}^M)} - \sqrt{2/3}pm^M$	
	$d\boldsymbol{\alpha} = (2/3) \langle L \rangle h(\boldsymbol{\alpha}^b - \boldsymbol{\alpha})$	
	$\boldsymbol{\alpha}^{b} = \sqrt{2/3} \left[ g(\theta) M \exp(-n^{b} \Psi) - m \right] \boldsymbol{n}$	$n^b$ bounding surface evolution parameter $M$ critical stress ratio
	$g(\theta) = 2c/[(1+c) - (1-c)\cos 3\theta]$	c extension-to-compression strength ratio
Plastic hardening	$L = (1/K_p)\partial f / \partial \boldsymbol{\sigma} : d\boldsymbol{\sigma}$	
8	$K_p = (2/3)ph(\boldsymbol{\alpha}^{\scriptscriptstyle D} - \boldsymbol{\alpha}) : \boldsymbol{n}$	
	$ \begin{array}{l} \boldsymbol{n} = (\boldsymbol{r} - \boldsymbol{\alpha})/\sqrt{2/3m} \\ \Psi = \boldsymbol{e} - \boldsymbol{e}_{c} \end{array} $	
	$h = \frac{b_0}{(\boldsymbol{\alpha} - \boldsymbol{\alpha}_{in}) : \boldsymbol{n}} \exp\left[\mu_0 \left(\frac{p}{p_{atm}}\right)^{0.5} \left(\frac{b^M}{b_{ref}}\right)^{w_1} \frac{1}{\eta^{w_2}}\right]$	$\mu_0$ , $w_1$ memory-hardening parameters $w_2$ cyclic stress ratio parameter
	$b_0 = G_0 h_0 (1 - c_h e) / \sqrt{p / p_{atm}}$	$h_0, c_h$ hardening parameters
	$b^{m} = (\mathbf{r}_{\alpha}^{m} - \boldsymbol{\alpha}) : \mathbf{n}$ $b = -(\mathbf{r}_{\alpha}^{b} - \mathbf{\tilde{r}}_{\alpha}^{b}) \cdot \mathbf{r}$	
	$b_{ref} = (\mathbf{a}^r - \mathbf{a}^r) : \mathbf{n}$ $\tilde{\mathbf{a}}_{b} = -\frac{\sqrt{2}/2}{2} [\mathbf{a}_{b}(\mathbf{a} + \mathbf{a}^r) \mathbf{M} \exp(-\mathbf{n}_{b}^{b} \mathbf{W}) - \mathbf{m}_{b}^{b} \mathbf{w}]$	
	$a^{m} = -\sqrt{2}/5[g(\theta + \pi)M\exp(-h^{2}\Psi) - m]\pi$	
	$\frac{\mathbf{r}_{\alpha} - \mathbf{u} + \sqrt{2/5(m^2 - m)n}}{dm^M - dm^M + dm^M}$	
	$dm^{M} = \sqrt{3/2} d\boldsymbol{\alpha}^{M} \cdot \boldsymbol{n}$	
M	$dm_{+}^{M} = -(m^{M}/\zeta) f_{shr} \langle b_{r}^{b} \rangle m_{+}^{M} \langle -d\varepsilon^{p} \rangle$	$\zeta$ memory surface shrinkage parameter
Memory surface	$F = \ln \left[ 1 +  m_{\perp}^{M}  / ( m_{\perp}^{M}  +  m_{\perp}^{M} )^{0.5} \right]$	S memory survive survivage Furameter
evolution	$b_r^b = (\boldsymbol{a}^b - \boldsymbol{a}) : \boldsymbol{n}$	
	$d\boldsymbol{\alpha}^{M} = (2/3) \left\langle L^{M} \right\rangle h^{M} (\boldsymbol{\alpha}^{b} - \boldsymbol{r}_{\alpha}^{M})$	
	$h^{M} = \frac{1}{2} \left[ \frac{b_{0}}{(\boldsymbol{r}^{M} - \boldsymbol{\alpha}_{i+}) \cdot \boldsymbol{n}} + \sqrt{\frac{3}{2}} \frac{m^{M} m_{+}^{M} \langle b_{r}^{b} \rangle f_{shr} \langle -D \rangle}{\zeta(\boldsymbol{\alpha}^{b} - \boldsymbol{r}^{M}) \cdot \boldsymbol{n}} \right]$	
	$de^{p} = \langle L \rangle R' = \langle L \rangle \{Bn - C [n^{2} - (1/3)L] \}$	
Deviatoric	$B = 1 + 3(1 - c)/(2c)g(\theta)\cos 3\theta$	
plastic flow	$C = 3\sqrt{3/2}(1-c)/cg(\theta)$	
	$d\varepsilon_{vol}^{p} = \langle L \rangle D$	
	$d = (\boldsymbol{\alpha}^d - \boldsymbol{\alpha}) : \boldsymbol{n}$	
	$D = A_d d$	
Volumetric	$A_d = A_0 \text{ (for } d \ge 0 \text{ and } b_d^M \le 0)$	$A_0$ 'intrinsic' dilatancy parameter
plastic flow	$A_d = A_0 \exp \left[ \beta_1 F \left( \frac{p}{p_{max}} \right)^{0.5} \right] g^k(\theta) \text{ (for } d \ge 0 \text{ and } \tilde{b}_d^M > 0)$	$\beta_1$ dilatancy parameter k dilatancy parameter
	$A_d = A_0 \exp\left[\beta_2 F\left(1 - \left(\frac{p}{p_{max}}\right)^{0.5}\right) \frac{d}{  \boldsymbol{a}^c  }\right] \frac{1}{g(\theta)} \text{ (for } d < 0)$	$\beta_2$ dilatancy parameter
	$\boldsymbol{\alpha}^{c} = \sqrt{2/3}(g(\theta)M - m)\boldsymbol{n}$	
	$\begin{bmatrix} \boldsymbol{\alpha}^{d} = \sqrt{2}/3 \left[ g(\theta) \boldsymbol{M} \exp(n^{d} \boldsymbol{\Psi}) - \boldsymbol{m} \right] \boldsymbol{n} \\ \tilde{\boldsymbol{h}}_{M}^{M} = (\tilde{\boldsymbol{\alpha}}^{d} - \tilde{\boldsymbol{r}}_{M}^{M}) : \boldsymbol{n} \end{bmatrix}$	$n^d$ dilatancy surface evolution parameter
	$\tilde{\boldsymbol{\alpha}}^{d} = -\sqrt{2/3} \left[ g(\theta + \pi) M \exp(n^{d} \Psi) - m \right] \boldsymbol{n}$	

## 430 DATA AVAILABILITY STATEMENT

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Some or all data, models, or code that support the findings of this study are available from the

432 corresponding author upon reasonable request.

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FEATURE	PARAMETER	VALUE
Flasticity	$G_0$	95
Liasticity	ν	0.05
	М	1.35
Critical	С	0.81
cifical	$\lambda_c$	0.055
State	<i>e</i> <sub>0</sub>	1.035
	ξ	0.36
Yield	т	0.01
Diastia	$h_0$	7.6
Modulus	C <sub>h</sub>	0.97
Modulus	$n^b$	1.2
Dilatanov	$A_0$	0.74
Dilatancy	$n^d$	1.79
Momory	$\mu_0$	82
surface	ζ	0.0005
Surface	β	4

**TABLE 1.** Parameters of Liu et al. (2019) model for the Karlsruhe fine sand tested by Wichtmann & Triantafyllidis (2016)

**TABLE 2.** Upgraded SANISAND-MS parameters for the Karlsruhe fine sand tested by Wichtmann& Triantafyllidis (2016)

FEATURE	PARAMETER	VALUE
Flocticity	$G_0$	95
Elasticity	ν	0.05
	М	1.35
Critical	С	0.81
cifical	$\lambda_c$	0.055
State	<i>e</i> <sub>0</sub>	1.035
	ξ	0.36
Yield	m	0.01
Plastic	$h_0$	7.6
Modulus	C <sub>h</sub>	0.97
Wiodulus	$n^b$	1.2
	$A_0$	0.74
	$n^d$	1.79
Dilatancy	$\beta_1$	4
	$\beta_2$	3.2
	k	2
	$\mu_0$	65
Memory	ζ	0.0005
surface	<i>w</i> <sub>1</sub>	2.5
	w <sub>2</sub>	1.5

FEATURE	PARAMETER	VALUE
Flasticity	$G_0$	110
Elasticity	ν	0.05
	М	1.27
Critical	С	0.712
state	$\lambda_c$	0.049
State	<i>e</i> <sub>0</sub>	0.845
	ξ	0.27
Yield	m	0.01
Plastic	$h_0$	5.95
Modulus	C <sub>h</sub>	1.01
Wiodulus	$n^b$	2
	$A_0$	1.06
	$n^d$	1.17
Dilatancy	$\beta_1$	1.9
	$\beta_2$	2.1
	k	1
	$\mu_0$	260
Memory	ζ	0.0001
surface	<i>w</i> <sub>1</sub>	2
	<i>w</i> <sub>2</sub>	0

**TABLE 3.** Upgraded SANISAND-MS parameters for the quartz sand tested by Wichtmann (2005)

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Fig. 1. SANISAND-MS loci in the deviatoric stress ratio plane.



(a) Memory surface expansion and translation

(**b**) Memory surface contraction

Fig. 2. Evolution of the memory surface



**Fig. 3.** Cyclic DSS simulations via SANISAND-MS. Simulation conditions:  $e_{in} = 0.812$  (initial void ratio),  $\sigma_v = 100$  kPa (effective vertical stress),  $\tau_{ampl} = \pm 20$  kPa (cyclic shear stress amplitude); cyclic parameters in the upgraded model:  $\mu_0 = 65$ ,  $\zeta = 0.0003$ ,  $w_1 = 2.5$ ,  $w_2 = 1.5$ , k = 2.



**Fig. 4.** Cyclic triaxial simulations on isotropically consolidated sand via SANISAND-MS. Simulation settings:  $e_{in} = 0.825$ ,  $p_{in} = 100$  kPa,  $q_{ampl} = 30$  kPa. Cyclic parameters in the upgraded model:  $\mu_0 = 65$ ,  $\zeta = 0.0003$ ,  $w_1 = 2.5$ ,  $w_2 = 1.5$ ,  $\beta_1 = 4.0$ ,  $\beta_2 = 3.2$ , k = 2.



Fig. 5. Calibration of  $\beta_1$ . Test/simulation settings and cyclic parameters are as in Fig.4b (Data from Wichtmann and Triantafyllidis 2016).





(a) triaxial test (Data from Wichtmann and Triantafyllidis 2016)

(b) upgraded SANISAND-MS simulations

**Fig. 6.** Calibration of  $\beta_2$  and k. Test/simulation settings:  $e_{in} = 0.8$ ,  $p_{in} = 200$  kPa,  $q_{ampl} = 200$  kPa. Cyclic parameters in the upgraded model:  $\mu_0 = 65$ ,  $\zeta = 0.0003$ ,  $w_1 = 2.5$ ,  $w_2 = 1.5$ ,  $\beta_1 = 4.0$ . Number of loading cycles after initial liquefaction N = 10.



**Fig. 7.** Performance of previous and upgraded SANISAND-MS (model parameters in Table 1 and Table 2, respectively) on pore pressure accumulation in isotropically consolidated sand under varying stress amplitude ratios  $\eta_{ampl}$ . Test/simulation settings: performed with an initial drained loading cycle,  $p_{in} = 300$  kPa,  $e_{in} = 0.846$  when  $\eta_{ampl} = 0.2$ ;  $e_{in} = 0.816$  when  $\eta_{ampl} = 0.3$  (Data from Wichtmann and Triantafyllidis 2016).



Fig. 8. Calibration of  $w_2$  based on the results of undrained cyclic triaxial tests on isotropically consolidated sand (Data from Wichtmann and Triantafyllidis 2016.



**Fig. 9.** Calibration of  $w_1$  and  $\mu_0$ . Test/simulation settings: performed with an initial drained loading cycle,  $e_{in} = 0.808$ ,  $p_{in} = 300$  kPa,  $\eta_{ampl} = 0.25$ . Cyclic parameters in the upgraded model:  $\beta_1 = 4.0$ ,  $\beta_2 = 3.2$ ,  $w_2 = 1.5$ , k = 2,  $\zeta = 0.0003$ .



**Fig. 10.** Pore pressure accumulation curves. Same test/simulation settings as in Fig.7. Comparison among experimental data (Wichtmann & Triantafyllidis, 2016), empirical fit (Chiaradonna et al., 2018) and SANISAND-MS simulations.



(e) SANISAND-MS vs experimental results: pore pressure accumulation predictions from Liu et al. (2019)'s formulation and upgraded model

**Fig. 11.** Influence of initial effective mean pressure on pore pressure accumulation in isotropically consolidated sand. Test/simulation settings: performed with an initial drained loading cycle,  $p_{in} = 100 \text{ kPa} (e_{in} = 0.798)$ , 200 kPa ( $e_{in} = 0.813$ ) and 300 kPa ( $e_{in} = 0.808$ ),  $\eta_{ampl} = 0.25$ . Comparison between experimental data (Wichtmann & Triantafyllidis, 2016) and SANISAND-MS simulations.



(e) SANISAND-MS vs experimental results: pore pressure accumulation predictions from Liu et al. (2019)'s formulation and upgraded model

**Fig. 12.** Influence of cyclic amplitude ratio  $\eta_{ampl}$  on undrained cyclic behaviour of isotropically consolidated sand. Test/simulation settings: performed with an initial drained loading cycle,  $e_{in} = 0.821, 0.798, 0.825$  for  $\eta_{ampl} = 0.2, 0.25$  and 0.3;  $p_{in} = 100$  kPa. Comparison between experimental data (Wichtmann & Triantafyllidis, 2016) and SANISAND-MS simulations.



(a) Experimental results: q-p response for  $e_{in} = 0.825$  and  $e_{in} = 0.759$ 



(b) Upgraded SANISAND-MS results: q-p response for  $e_{in} = 0.825$  and  $e_{in} = 0.759$ 



(c) Experimental results:  $q \cdot \varepsilon_a$  response for  $e_{in} = 0.825$  and  $e_{in} = 0.759$ 



(d) Upgraded SANISAND-MS results:  $q \cdot \varepsilon_a$  response for  $e_{in} = 0.825$  and  $e_{in} = 0.759$ 



(e) SANISAND-MS vs experimental results: pore pressure accumulation predictions from Liu et al. (2019)'s formulation and upgraded model

**Fig. 13.** Influence of initial relative density on pore pressure accumulation in isotropically consolidated sand. Test/simulation settings: performed with an initial drained loading cycle, mediumdense sand ( $e_{in} = 0.825$ ) and dense sand ( $e_{in} = 0.759$ ),  $p_{in} = 100$  kPa,  $\eta_{ampl} = 0.3$ . Comparison between experimental data (Wichtmann & Triantafyllidis, 2016) and SANISAND-MS simulations. 45 Liu et al., July 14, 2020



(c) pore pressure generation

**Fig. 14.** Relative effect of cyclic stress amplitude ratio  $\eta_{ampl}$  and initial average stress ratio  $\eta_{ave}$  on the undrained effective stress path in anisotropically consolidated sand. Test/simulation settings: performed with an initial drained loading cycle,  $p_{in} = 200$  kPa,  $q_{ampl} = 60$  kPa (Data from Wichtmann and Triantafyllidis 2016).



**Fig. 15.** Effect of drained cyclic pre-loading on the undrained cyclic triaxial response of the quartz sand (Wichtmann, 2005) – isotropically consolidated sand. Test/simulation settings:  $e_{in} = 0.678$ ,  $p_{in} = 100$  kPa, cyclic stress amplitude during undrained loading:  $q_{ampl} = 45$  kPa.