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# Effect of CaO content in raw material on the mineral composition of ferric-rich sulfoaluminate clinker

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# 9 Abstract

10 Ferric-rich calcium sulfoaluminate (FR-CSA) cement is an eco-friendly cement. Fe<sub>2</sub>O<sub>3</sub> exists in 11 different minerals of FR-CSA clinker, e.g., Ca<sub>4</sub>Al<sub>2</sub>Fe<sub>2</sub>O<sub>10</sub> (C<sub>4</sub>AF), Ca<sub>2</sub>Fe<sub>2</sub>O<sub>5</sub> (C<sub>2</sub>F), and Ca<sub>4</sub>Al<sub>6</sub>. 12  $_{2y}$ Fe<sub>2y</sub>SO<sub>16</sub> (C<sub>4</sub>A<sub>3y</sub>F<sub>y</sub>S̄). The mineral composition depends on the chemical composition of the raw 13 materials and significantly determines the reactivity of FR-CSA cement. To optimize the phase 14 composition of the FR-CSA clinker, chemical reagent raw mixtures with different amounts of CaO 15 were used to prepare the FR-CSA clinker. X-ray diffraction (XRD) analysis, Rietveld quantitative 16 phase analysis (RQPA), Fourier Transform Infrared spectroscopy (FT-IR), and scanning electron 17 microscopy/energy-dispersive spectroscopy (SEM/EDS) were used to identify the mineralogical 18 conditions of the FR-CSA clinker. The results indicated that the amounts of CaO in raw materials 19 greatly affected the iron-bearing phase formation in the FR-CSA clinker. With decreasing CaO content 20 involved in calcination reaction, the amounts of Fe<sub>2</sub>O<sub>3</sub> incorporated in C<sub>4</sub>A<sub>3-x</sub> $F_x\overline{S}$  increased up to 17.72 21 wt.% (where x = 0.36). The findings make it possible to optimize the mineral composition of the FR-22 CSA clinker by changing the CaO content in raw materials. Furthermore, low CaO content in the raw 23 material is beneficial to the formation of  $C_4A_{3,r}F_s\bar{S}$ , which enables the use of solid wastes containing 24 low calcium for producing FR-CSA cement.

Keywords: Ferric-rich sulfoaluminate cement, CaO, Optimization, Phase composition, Iron-bearing
 phase

#### 27

# 28 1. Introduction

29 Calcium sulfoaluminate (CSA) cement is a new type of cement first developed in China 40 years 30 ago. It exhibits some great performances, such as high early strength, rapid setting, and shrinkage-31 compensating, et al. for certain applications [1-3]. The major phases of CSA clinker are ye'elimite 32  $(Ca_4Al_6SO_{16}, C_4A_3\overline{S})$ , belite  $(Ca_2SiO_4, C_2S)$ , and the minor phases are anhydrite  $(CaSO_4, C\overline{S})$ , 33 brownmillerite (Ca<sub>4</sub>Al<sub>2</sub>Fe<sub>2</sub>O<sub>10</sub>, C<sub>4</sub>AF), and Ca<sub>2</sub>Fe<sub>2</sub>O<sub>5</sub> (C<sub>2</sub>F), et al[4, 5]. C<sub>4</sub>A<sub>3</sub>S in CSA contains smaller 34 amount of CaO than the alite (Ca<sub>3</sub>SiO<sub>5</sub>, C<sub>3</sub>S) in Portland cement (PC). Therefore, the production of 35 CSA clinker requires lower amount of limestone consumption compared with Portland cement (PC) 36 production, and reduces the  $CO_2$  emission [6]. Meanwhile, the preparation temperature of CSA clinker 37 ranges from 1200-1300 °C, which is 200 °C lower than the calcination temperature of Portland cement 38 clinker, reducing energy consumption during the calcination process[7, 8]. Besides, the CSA clinkers 39 are easier to grind due to their high porosity, which further reduces energy consumption during the 40 grinding process. Therefore, the characters of energy-saving, low CO<sub>2</sub> emission, and good mechanical 41 performance, make CSA cement as an ideal alternative to PC in certain applications.

However, the utilization of CSA cement is still limited. The main obstacle to the application of CSA cement on a massive scale is its high cost of raw materials, which requires expensive natural bauxite. Therefore, the consumption of bauxite, which contains high amount of Al<sub>2</sub>O<sub>3</sub>, should be

1 minimized. Fe<sub>2</sub>O<sub>3</sub> is usually used as an alternative component to Al<sub>2</sub>O<sub>3</sub> during the production process 2 of CSA clinker, resulting in the FR-CSA cement. Note that the required amount of Fe<sub>2</sub>O<sub>3</sub> in the FR-3 CSA clinker ranges from 5 to 13 wt.%, while only 1 to 3 wt.% of Fe<sub>2</sub>O<sub>3</sub> exist in the CSA clinker [9]. In 4 addition to C<sub>2</sub>F and C<sub>4</sub>AF, the Fe<sub>2</sub>O<sub>3</sub> can also be incorporated in C<sub>4</sub>A<sub>3</sub> $\bar{S}$  to form C<sub>4</sub>A<sub>3-x</sub>F<sub>x</sub> $\bar{S}$  in the FR-5 CSA clinker, where the value of x represents the amount of Fe<sub>2</sub>O<sub>3</sub> incorporated in C<sub>4</sub>A<sub>3</sub> $\overline{S}$  [10, 11]. 6 Similar to the reaction of  $C_4A_3\overline{S}$  with water,  $C_4A_{3x}F_x\overline{S}$  can react with water and form ettringite, which 7 is conducive to the development of compressive strength [9]. Thus,  $Fe_2O_3$  can be used as a partial 8 replacement of Al<sub>2</sub>O<sub>3</sub> to reduce the consumption of aluminum source. With the reduced requirement of 9 the consumption of aluminum source, some aluminum-bearing solid wastes or low-grade Al<sub>2</sub>O<sub>3</sub> 10 minerals can be used, and the cost of FR-CSA cement can be reduced.

11 During the calcination process of the FR-CSA clinker, the distributions of  $Fe_2O_3$  in iron-bearing 12 phases are affected by the chemical composition of the raw material. And the iron-bearing phase 13 composition influences the mechanical properties of the FR-CSA cement [12, 13]. To optimize the 14 iron-bearing phase composition of FR-CSA clinker, it is important to know the effect of raw material 15 composition on the iron-bearing phase formation in the FR-CSA clinker. Many researchers have 16 investigated the effect of raw material on the phase composition of the FR-CSA clinker. Y. Huang et al. 17 concluded that the addition of  $Fe_2O_3$  could promote higher  $Fe_2O_3$  incorporation into CSA clinker, 18 forming C<sub>4</sub>A<sub>27</sub>F<sub>0.3</sub> $\overline{S}$  [14]. Dun Chen et al. studied the effect of Fe<sub>2</sub>O<sub>3</sub> content on the incorporation level 19 of Fe<sub>2</sub>O<sub>3</sub> into C<sub>4</sub>A<sub>3-x</sub>F<sub>x</sub> $\overline{S}$  and concluded that 22.31 wt.% Fe<sub>2</sub>O<sub>3</sub> reached maximum incorporation content 20 into  $C_4A_{3-x}F_x\overline{S}[10]$ . Idrissi M. et al. identified the mineralogical conditions of iron inclusion during the 21 formation of  $C_4A_3\overline{S}$  made from different raw materials. The maximum fraction of Fe<sub>2</sub>O<sub>3</sub> in the  $C_4A_3\overline{S}$ 22 was 21.5 wt.%[15]. Bruno Touzo et al. studied the phase equilibria in the CaO-Al<sub>2</sub>O<sub>3</sub>-Fe<sub>2</sub>O<sub>3</sub>-SO<sub>3</sub> 23 system at 1325  $^{\circ}$ C and got that the value of x increased to a plateau of 0.34[16]. All the aforementioned 24 researches were focused on the effect of Fe<sub>2</sub>O<sub>3</sub>/(Fe<sub>2</sub>O<sub>3</sub>+Al<sub>2</sub>O<sub>3</sub>) ratio in CaO-Al<sub>2</sub>O<sub>3</sub>-Fe<sub>2</sub>O<sub>3</sub>-SO<sub>3</sub> system. 25 However, CaO is a primary content in the FR-CSA clinker. During the calcination of FR-CSA clinker, 26 CaO almost participates in all the phase formation reactions and distributes in all minerals according to 27 Eqs. 1–8[17]. Thus, CaO content is a critical parameter for the phase composition in FR-CSA clinker.

28 Bruno Touzo et al. also put forward that Fe<sub>2</sub>O<sub>3</sub>/(Fe<sub>2</sub>O<sub>3</sub>+Al<sub>2</sub>O<sub>3</sub>) was not the only parameter 29 influencing the value of x. The CaO content and the content of other elements would shift the 30 equilibrium of the reactions during calcination. C. Ren et al. and Y. Shen et al. prepared the CSA 31 clinker from solid wastes and demonstrated that the CaO or CaSO<sub>4</sub> content affected the formation of 32  $C_2AS$  and  $C_4A_3\overline{S}$  in CSA clinker[18, 19]. However, all the researches were about the effect of CaO 33 content on phase formation or  $C_4A_3\overline{S}$  in CSA clinker. Investigation on the effect of CaO content on the 34 formation of iron-bearing phases in FR-CSA clinker is still limited. To optimize the iron-bearing phase 35 composition of the FR-CSA clinker, it is necessary to investigate the effect of CaO content on the iron-36 bearing phase formation in FR-CSA clinker.

37 
$$Ca(OH)_2 \xrightarrow{500 \circ C} CaO+H_2O$$
 Eq. 1

$$CaCO_3 \xrightarrow{800-850 \ ^{\circ}C} CO_2 + CaO$$
 Eq. 2

39 
$$2\text{CaO+Al}_2\text{O}_3 + \text{SiO}_2 \xrightarrow{900-950 \,^\circ\text{C}} \text{C}_2\text{AS}$$
 Eq. 3

40 
$$3CaO+3Al_2O_3+CaSO_4 \xrightarrow{950-1000 \circ C} C_4A_3\overline{S}$$
 Eq. 4

$$3CaO+3C_2AS+CaSO_4 \xrightarrow{1050-1150 \ ^{\circ}C} C_4A_3\overline{S}+3C_2S$$
 Eq. 5

$$2\text{CaSO}_4 \xrightarrow{1170 \text{ °C}} \text{SO}_2 + \text{CaO} + \text{O}_2$$
 Eq. 6

1

$$2\text{CaO}+\text{Fe}_2\text{O}_3 \xrightarrow{1000-1100 \,^\circ\text{C}} \text{C}_2\text{F}$$
 Eq. 7

4

$$2\text{CaO}+\text{C}_2\text{F}+\text{Al}_2\text{O}_3 \xrightarrow{1150-1200 \ ^{\circ}\text{C}} \text{C}_4\text{AF}$$
 Eq. 8

5 The aim of this work is to investigate the feasibility of optimizing the phase composition of FR-6 CSA clinker by changing the CaO content in raw materials. In our research, the FR-CSA clinkers were 7 prepared from chemical reagents, in which different amount of  $Ca(OH)_2$  and  $CaSO_4$  were added to 8 change the CaO content, respectively. X-ray diffraction (XRD) analysis, Rietveld quantitative phase 9 analyses (RQPA), and Fourier Transform Infrared spectroscopy (FT-IR) were used to identify and 10 quantify the phase composition, respectively. Scanning electron microscopy/energy-dispersive 11 spectroscopy (SEM/EDS) was used to detect the distribution of  $Fe_2O_3$  in iron-bearing phases. Along 12 with the variation of iron-bearing phase compositions, the incorporated content of Fe<sub>2</sub>O<sub>3</sub> in C<sub>4</sub>A<sub>3-x</sub>F<sub>x</sub> $\bar{x}$ 13 was calculated. The effective utilizations of Fe<sub>2</sub>O<sub>3</sub> and Al<sub>2</sub>O<sub>3</sub> in the FR-CSA clinker were also 14 calculated according to the distribution of Fe<sub>2</sub>O<sub>3</sub> and Al<sub>2</sub>O<sub>3</sub> in different phases. Finally, the clinkers 15 with different C<sub>4</sub>AF designed and with different calcination process were also got to certify the effect 16 of CaO content in raw materials.

17

### 18 2. Experiment

19 2.1 Preparation of FR-CSA clinker

20 2.1.1 Raw materials

The chemical pure reagents (Ca(OH)<sub>2</sub>, Al<sub>2</sub>O<sub>3</sub>, CaSO<sub>4</sub>, Fe<sub>2</sub>O<sub>3</sub>, and SiO<sub>2</sub>), which were supplied by Sinopharm Group Chemical Reagent Co., Ltd., Shanghai, China, were used as raw materials to prepare the FR-CSA clinker. Among them, Ca(OH)<sub>2</sub> was used as the source of CaO. CaSO<sub>4</sub> was not only used as the source of CaO but also was the source of SO<sub>3</sub> during the calcination process of the FR-CSA clinker.

The FR-CSA clinker was synthesized from chemical reagent raw materials based on stoichiometric amount. The ratios of raw materials were calculated according to the Bogue method, which was adapted for CSA clinker by assuming phase assemblage of  $C_2S$ ,  $C_4A_3\overline{S}$ ,  $C_4AF$ , and  $CaSO_4$ as given in Eqs. 9-12[20]:

30 
$$C_4AF\% = (F\%) \times (\frac{M_{C_4AF}}{M_F})$$
 Eq. 9

31 
$$C_2 S\% = (S\%) \times (\frac{M_{C_2 S}}{M_S})$$
 Eq. 10

32 
$$C_4 A_3 \overline{S}\% = \left(A\% - C_4 A F\% \times \frac{M_A}{M_{C_4 A F}}\right) \times \frac{M_{C_4 A_3} \overline{S}}{3 \times M_A}$$
Eq. 11

33 
$$C\overline{S}\% = \left(\overline{S}\% - C_4 A_3 \overline{S}\% \times \frac{M_{\overline{S}}}{M_{C_4 A_3 \overline{S}}}\right) \times \frac{M_{C\overline{S}}}{M_{\overline{S}}}$$
Eq. 12

The CaO content in the FR-CSA clinkers was termed as the alkalinity modulus ( $C_m$ ) and defined using Eq. 13[9]. The value of *x* indicates the degree to which CaO in the raw material satisfies the reaction with SiO<sub>2</sub>, Al<sub>2</sub>O<sub>3</sub>, and Fe<sub>2</sub>O<sub>3</sub> to form useful minerals (i.e., C<sub>4</sub>A<sub>3</sub> $\overline{S}$ , C<sub>4</sub>AF, and C<sub>2</sub>S) in the CSA 1 clinker. It is usually lower than 1.00 to avoid the existence of free calcium oxide (f-CaO) in the CSA

2 clinker. The C<sub>m</sub> is calculated as follows.

$$C_{m} = \frac{CaO-0.7TiO_{2}}{1.87SiO_{2}+0.73(Al_{2}O_{3}-0.64Fe_{2}O_{3})+1.40Fe_{2}O_{3}}$$
Eq. 13

3 According to Table 1 and Eq. 9-12, five FR-CSA clinkers with different CaO content were 4 designed and termed as C<sub>0.90</sub>, C<sub>0.95</sub>, C<sub>1.00</sub>, C<sub>1.05</sub>, and C<sub>1.10</sub>, corresponding to the C<sub>m</sub> values of 0.90, 0.95, 5 1.00, 1.05, and 1.10, respectively. The targeted phase ratio of  $C_4A_3\overline{S}$ :  $C_4AF$ :  $C_2S$ :  $C_3SO_4$  was 40 %: 6 20 %: 30 %: 10 % by mass. Furthermore, to verify the results from the above raw mixture, as shown in 7 Table 2, another raw mixture that targeted phase ratio of C<sub>4</sub>A<sub>3</sub> $\overline{S}$ : C<sub>4</sub>AF: C<sub>2</sub>S: CaSO<sub>4</sub> was 30 %: 30 %: 8 30 %: 10 % by mass was also designed and termed as C<sub>0.90-1</sub>, C<sub>0.95-1</sub>, C<sub>1.00-1</sub>, C<sub>1.05-1</sub>, and C<sub>1.10-1</sub>.

	Variation		Proportions (wt.%)						
_	variation	C <sub>0.90</sub>	C <sub>0.95</sub>	C <sub>1.00</sub>	C <sub>1.05</sub>	C <sub>1.10</sub>			
	$C_4A_3\overline{S}$	40	40	40	40	40			
	$C_2S$	30	30	30	30	30			
FR-CSA clinker	$C_4AF$	20	20	20	20	20			
	$CaSO_4$	10	10	10	10	10			
	$\mathbf{C}_{\mathbf{m}}$	0.90	0.95	1.00	1.05	1.10			
	CaO	31.3	33.5	35.7	38.0	40.2			
	SiO <sub>2</sub>	10.7	10.7	10.7	10.7	10.7			
Raw material	CaSO <sub>4</sub>	21.7	21.7	21.7	21.7	21.7			
	$Al_2O_3$	24.9	24.9	24.9	24.9	24.9			
	Fe <sub>2</sub> O <sub>3</sub>	6.8	6.8	6.8	6.8	6.8			

Table 1 Chemical composition in raw mixture and targeted phase compositions for FR-CSA clinkers

10 In addition, CaO exists in the form of  $Ca(OH)_2$  and  $CaSO_4$ . The decomposition temperature of 11 Ca(OH)<sub>2</sub> is below 900 °C, which is lower than the formation temperature of C<sub>4</sub>A<sub>3</sub> $\bar{S}$ , C<sub>2</sub>S, C<sub>4</sub>AF, and 12  $C_2AS$ . However, the decomposition temperature of  $CaSO_4$  is higher than 1170 °C. Thus, the amount of 13 CaSO<sub>4</sub> further affects the formation of minerals in FR-CSA clinker, which is not the same as the effect 14 of CaO content. To explore the effect of CaO in the form of CaSO<sub>4</sub> on the phase formation of FR-CSA 15 clinker, as shown in Table 2, another five clinkers with different CaSO4 contents but same Cm, termed 16 as S<sub>0</sub>, S<sub>5</sub>, S<sub>10</sub>, S<sub>15</sub>, and S<sub>20</sub>, were designed. The extra content of CaSO<sub>4</sub> was 0, 5, 10, 15, and 20 wt.% in 17 FR-CSA clinkers, respectively. The value of C<sub>m</sub> was 1.00 and Al<sub>2</sub>O<sub>3</sub>/Fe<sub>2</sub>O<sub>3</sub> ratios were same for all the 18 clinkers. 19

Table 2 Targeted phase compositions of the FR-CSA clinkers

Targeted	Proportions (wt.%)									
phases/Variation	C <sub>0.90-1</sub>	C <sub>0.95-1</sub>	C <sub>1.00-1</sub>	C <sub>1.05-1</sub>	C <sub>1.10-1</sub>	$\mathbf{S}_0$	<b>S</b> <sub>5</sub>	<b>S</b> <sub>10</sub>	<b>S</b> <sub>15</sub>	$\mathbf{S}_{20}$
$C_4 A_3 \overline{S}$	30	30	30	30	30	50	47.5	45	42.5	40
$C_2S$	30	30	30	30	30	30	28.5	27	25.5	24
$C_4AF$	30	30	30	30	30	20	19	18	17	16
CaSO <sub>4</sub>	10	10	10	10	10	0	5	10	15	20
C <sub>m</sub>	0.90	0.95	1.00	1.05	1.10	1.00	1.00	1.00	1.00	1.00

20 2.1.2 Preparation process

- 1 The raw materials were dried at  $105 \pm 5$  °C until constant weight. Next, raw materials with a 2 certain ratio were ground and sieved through a 200-mesh sieve to obtain the desired raw mixture. Then 3 the raw mixture was calcined at 1250 °C for 30 min, and rapidly cooled at 20 °C to obtain FR-CSA
- 4 clinkers. Finally, it was ground and sieved through a 200-mesh sieve for the following analyses.
- 5 2.2 Characterization methods
- 6 The f-CaO content in the FR-CSA clinkers was determined by the phenyl formic acid titration 7 method in accordance with the Chinese standard GB/T 176-2017[21].

8 The mineral composition of different clinkers was identified using an X-ray diffractometer (Aeris, 9 MalvernPanalytical, Netherlands) with Cu-Ka radiation at 40 kV and 15 mA. The measurement time 10 was 0.5 h per pattern with a scanning speed of 0.6 °/min during the 2 $\theta$  range of 10–50 ° for qualitative 11 analysis. RQPA was applied to perform a quantitative phase analysis using X'Pert HighScore Plus 12 software. The crystallographic structures of phases involved in RQPA were listed in Table 3. Except 13 for  $C_4A_{3-x}F_x\overline{S}$ , which was described using a refined crystallographic structure, other minerals remained 14 the same with the reference phases. The refinement parameters were in order as follows: emission 15 profile, background coefficients, instrument parameters, zero error, Lorentz polarization factor, unit 16 cell parameters and preferred orientation coefficient.

17

Phase Space Group ICSD code Phase Space Group ICSD code 80361 I-43m Pcc2 9560 C<sub>4</sub>A<sub>3</sub>S-o C<sub>4</sub>A<sub>3</sub>S-c C<sub>2</sub>S-be P121/nl 79551 C<sub>2</sub>S-al' Pnma 81097  $C_4AF$ Ibm2 9197 CaSO<sub>4</sub> Bmmb 16382  $C_2AS$ P-421m 87144 Pnma 85123  $C_5S_2S$ C<sub>2</sub>F Pnma 15059 Si Fd-3mS 29287

Table 3 Crystallographic structures of phases involved in the RQPA

18 The FT-IR spectra of the FR-CSA clinkers were obtained using a Nicolet iS (Thermo Fisher, 19 USA). For the analysis of the different FR-CSA clinkers, 2 mg powder samples were obtained by 20 grinding the clinker in 400 mg of KBr. FT-IR scans were performed at frequencies from 400 cm<sup>-1</sup> to 21 1200 cm<sup>-1</sup> with a resolution of 4 cm<sup>-1</sup> at 20 °C. The results were analyzed using OMNIC software.

22 SEM (JSM-IT500HR, JEOL, Japan) with the 5 kV accelerated voltage was used to scan the 23 clinker particles after Au coating treatment. SEM and EDS-mapping pictures were got to investigate 24 the distribution of elements in the iron-bearing mineral grains.

26 3. Results and discussion

27

25

3.1 Analysis of FR-CSA clinkers with different amount of CaO in raw material 28

3.1.1 Mineralogical characterization of different FR-CSA clinkers

29 Fig. 1 shows the XRD patterns of the different FR-CSA clinkers. The main phases of clinkers are 30  $C_4A_3\overline{S}$ ,  $C_2S$ ,  $CaSO_4$ ,  $C_4AF$ , and  $C_2AS$ . The characteristic peak intensities of  $C_4AF$  and  $CaSO_4$  increase 31 dramatically with increasing Ca(OH)<sub>2</sub>. The peaks of C<sub>2</sub>AS are only visible in clinker C<sub>0.90</sub>. Moreover, a 32 slight d-spaces offset for the highest peak of  $C_4A_3\overline{S}$  (the cell parameter of  $C_4A_3\overline{S}$ -c and  $C_4A_3\overline{S}$ -o are (2) 33 1 1) and (0 2 2), respectively) are observed. The d-spaces of clinkers  $C_{0.90}$ ,  $C_{0.95}$ ,  $C_{1.00}$ ,  $C_{1.05}$ , and  $C_{1.10}$ 34 are 3.7627, 3.7621, 3.7612, 3.7596, and 3.7587 Å, respectively. With increasing CaO content in raw 35 material, the d-spaces decrease from 3.7627 to 3.7587 Å. It is speculated that some  $Fe_2O_3$  is 36 incorporated in C<sub>4</sub>A<sub>3</sub> $\overline{S}$  mineral to form C<sub>4</sub>A<sub>3-x</sub>F<sub>x</sub> $\overline{S}$  in the FR-CSA clinker. The radius of Fe<sup>3+</sup> (r=0.069 nm) is larger than that of  $Al^{3+}$  (r=0.053 nm)[22]. Fe<sup>3+</sup> occupies the octahedral position, whereas  $Al^{3+}$ 37

1 occupies the tetrahedral position in the  $C_4A_{3-x}F_x\overline{S}$ . Thus, with increasing CaO content, the content of 2 Fe<sub>2</sub>O<sub>3</sub> incorporated into  $C_4A_{3-x}F_x\overline{S}$  may decreases, which leads to a decrease in the volume of the unit 3 cell structure of  $C_4A_{3-x}F_x\overline{S}$  phase and a shift of the d-space of the main peak to a smaller value.

In addition, the interplanar spacing of  $C_4A_{3-x}F_x\bar{S}$  should be increase after Fe<sub>2</sub>O<sub>3</sub> incorporated in C<sub>4</sub>A<sub>3-x</sub>F<sub>x</sub> $\bar{S}$ , but the variations of the unit cell are inconsistent in different researches. The unit cell of different FR-CSA clinkers are obtained by the Rietveld refinement. As shown in Table 3, the reference crystal structures are C<sub>4</sub>A<sub>3</sub> $\bar{S}$ -o and C<sub>4</sub>A<sub>3</sub> $\bar{S}$ -c. After Rietveld refinement, the unit cells of C<sub>4</sub>A<sub>3-x</sub>F<sub>x</sub> $\bar{S}$  in different FR-CSA clinkers are shown in Table 4. With the increase of CaO content in raw material, the lattice constants of cubic cell C<sub>4</sub>A<sub>3-x</sub>F<sub>x</sub> $\bar{S}$  decrease gradually. For the orthorhombic cell C<sub>4</sub>A<sub>3-x</sub>F<sub>x</sub> $\bar{S}$ , the

10 variation of cell parameters a and b are irregular, but the parameter c decreases gradually. These 11 phenomena show a good correlation with other reported results [10, 15, 23].

12

Table 4 Refined unit cell for  $C_4A_{3-x}F_x\overline{S}$ 

		$C_4A_{3-x}F_x\overline{S}$ -c			$C_4A_{3-x}F_x\overline{S}$ -o			
	а	b	с	а	b	с		
C <sub>0.90</sub>	9.210	9.210	9.210	12.957	13.001	9.220		
C <sub>0.95</sub>	9.209	9.209	9.209	12.960	13.000	9.207		
C <sub>1.00</sub>	9.207	9.207	9.207	13.087	13.036	9.170		
C <sub>1.05</sub>	9.207	9.207	9.207	13.071	13.030	9.168		
C <sub>1,10</sub>	9.204	9.204	9.204	13.068	13.029	9.162		







Fig. 1 XRD patterns of the FR-CSA clinkers with different value of Cm in raw materials





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Fig. 2 Rietveld refinement plots of clinker  $C_{1.00}$ . The black line indicates the experimental scan, the red line indicates the calculation pattern, the middle gray line represents the difference curve, and the short line at the bottom indicates the peak positions of different phases

5 The mineral compositions of the clinkers were analyzed via RQPA using HighScore Plus software 6 [24, 25]. Fig. 2 shows the Rietveld refinement plots of clinker  $C_{1.00}$ . The weighted profile R-factor ( $R_{wp}$ ) 7 is 8.161%.  $C_4A_3\bar{S}$  exists in the orthorhombic and cubic crystal structure in the clinker.  $C_2S$  exists in the 8 form of beta-C<sub>2</sub>S and alpha'-C<sub>2</sub>S. The amounts of CaSO<sub>4</sub>, C<sub>4</sub>AF, and C<sub>2</sub>F in clinker C<sub>1.00</sub> are also 9 calculated, as shown in Table 5. Furthermore, the amounts of different phases in clinkers C<sub>0.90</sub>, C<sub>0.95</sub>, 10  $C_{1.05}$ , and  $C_{1.10}$  are also calculated according to the RQPA results and presented in Table 5. 11 Theoretically, the targeted phases of clinkers include  $C_4A_3\overline{S}$ ,  $C_2S$ ,  $C_4AF$ , and  $CaSO_4$ . However, the 12 phase formation is complicate and uncontrollable in the CaO-Al<sub>2</sub>O<sub>3</sub>-Fe<sub>2</sub>O<sub>3</sub>-SO<sub>3</sub>-SiO<sub>2</sub> system [26]. 13 Thus, small amounts of  $C_2F$ ,  $C_5S_2\overline{S}$ , and  $C_2AS$  are also formed in the FR-CSA clinkers, resulting in the 14 difference between the theoretical and quantified phases. With the increase of the amount of CaO in 15 raw material, the contents of  $C_4A_3\overline{S}$  and  $C_2AS$  decrease, while the amount of  $C_4AF$ ,  $CaSO_4$ , and  $C_2F$ 16 increases.

17 For the clinker C<sub>0.90</sub>, the amount of CaO is too small to exhaust Al<sub>2</sub>O<sub>3</sub>, Fe<sub>2</sub>O<sub>3</sub>, and SiO<sub>2</sub> during the 18 calcination process. When CaO in the forms of Ca(OH)<sub>2</sub> and CaCO<sub>3</sub> are completely consumed, 19 additional CaO from decomposing of CaSO<sub>4</sub> is required to continue to react with Al<sub>2</sub>O<sub>3</sub>, Fe<sub>2</sub>O<sub>3</sub> or SiO<sub>2</sub>, 20 which promotes the decomposition of CaSO<sub>4</sub>. However, for the clinker  $C_{1,10}$ , the amount of CaO is 21 enough to react with Al<sub>2</sub>O<sub>3</sub>, Fe<sub>2</sub>O<sub>3</sub>, and SiO<sub>2</sub> according to the stoichiometric raw mixture, with only 22 minor decomposition of CaSO<sub>4</sub>. Thus, the amount of CaSO<sub>4</sub> in FR-CSA clinker increases with 23 increasing in CaO. In addition, although the values of  $C_m$  are higher than 1.00 for clinkers  $C_{1.05}$  and 24  $C_{1.10}$ , but f-CaO does not exist in all the clinkers. Therefore, the properties of FR-CSA cement will not 25 be influenced by f-CaO.



Table 5 Phase compositions of different clinkers (wt.%) (Theo.: theoretical phases calculated using the Bogue equations.

Quant.: phases quantified via Rietveld refinement)

Clinkers	Theo	Quant.					
Mineral	Theo.	C <sub>0.90</sub>	C <sub>0.95</sub>	C <sub>1.00</sub>	C <sub>1.05</sub>	C <sub>1.10</sub>	
$C_4A_3\overline{S}$ -o	—	14.89	16.21	15.65	16	15.23	
$C_4A_3\overline{S}$ -c	—	40.65	37.94	35.98	31.28	28.15	

$C_4A_3\overline{S}$	40	55.84	54.15	51.63	46.98	43.38
C <sub>2</sub> S-beta	—	25.04	17.2	10.51	11.47	27.77
C <sub>2</sub> S-alpha'	—	8.71	15.75	22.11	20.17	4.5
$C_2S$	30	33.75	32.95	32.62	31.64	32.27
CaSO <sub>4</sub>	10	0.54	1.91	4.29	7.76	8.47
$C_2AS$	—	5.09	2.79	1.94	1.41	0.95
$C_5S_2\overline{S}$	_	1.16	1.79	1.74	2.41	2.99
C <sub>4</sub> AF	20	2.60	4.93	6.28	7.92	9.35
$C_2F$	—	1.02	1.48	1.5	1.88	2.59

3.1.2 Incorporation levels of Fe<sub>2</sub>O<sub>3</sub> in C<sub>4</sub>A<sub>3-x</sub> $F_x\overline{S}$ 

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2 The experimentally quantified  $C_4A_3\bar{S}$  content, theoretical calculated  $C_4A_3\bar{S}$  content, and the 3 differences between them are shown in Fig. 3. The theoretical content of  $C_4A_3\overline{S}$  in different clinkers are 4 40 wt.%, but the experimental results are different. With increasing amount of CaO in the raw materials, 5 the amount determined by experiments and its deviation from theoretical amounts of  $C_4A_3\bar{S}$  phase 6 decrease significantly. Since the cell parameters of  $C_4A_3\bar{S}$  and  $C_4A_{3-x}F_x\bar{S}$  phases are similar, the  $C_4A_3\bar{S}$ 7 content determined by experiments actually consists of the  $C_4A_{3-x}F_x\overline{S}$  and pure  $C_4A_3\overline{S}$ . When some 8 amount of Fe<sub>2</sub>O<sub>3</sub> are incorporated in C<sub>4</sub>A<sub>3</sub> $\overline{S}$  to form C<sub>4</sub>A<sub>3-x</sub>F<sub>x</sub> $\overline{S}$ , different incorporation levels lead to the 9 variation of the amount of  $C_4A_3\overline{S}$  phase determined by experiments. For the Fe<sub>2</sub>O<sub>3</sub> in the FR-CSA 10 clinker, in addition to forming C<sub>4</sub>AF and C<sub>2</sub>F, the remaining Fe<sub>2</sub>O<sub>3</sub> replace Al<sub>2</sub>O<sub>3</sub> in C<sub>4</sub>A<sub>3-x</sub> $F_x\overline{S}$ . Hence, 11 according to previous research [27], the amount of  $Fe_2O_3$  involved in the substitution reaction is 12 calculated using the  $Fe_2O_3$  content in C<sub>4</sub>AF and C<sub>2</sub>F. The substitution amount of  $Fe_2O_3$  and the values 13 of x in C<sub>4</sub>A<sub>3-x</sub> $F_x\bar{S}$  are calculated and presented in Table 6. With increasing CaO content in raw material, 14 the incorporation levels of Fe<sub>2</sub>O<sub>3</sub> in C<sub>4</sub>A<sub>3-x</sub>F<sub>x</sub> $\overline{S}$  decrease from 17.41 wt.% to 8.89 wt.%. The values of x 15 in  $C_4A_{3-x}F_x\overline{S}$  decrease from 0.36 to 0.18 correspondingly.

16

Table 6 Substitution amount of $Fe_2O_3$ in $C_4A_{3-x}F_xS$									
Clinkers	Substitution amount (wt.%)	X							
C <sub>0.90</sub>	17.72	0.36							
C <sub>0.95</sub>	14.66	0.30							
C <sub>1.00</sub>	13.69	0.28							
C <sub>1.05</sub>	11.92	0.24							
C <sub>1.10</sub>	8.96	0.18							
- 00 - 00 - 05 - 05 - 05 - 05 - 05 - 05		2 Quant. 18 16 16 16 14 12 Ling and 18 16 14 12 Ling and 18 18 18 16 16 14 18 18 16 16 18 18 18 18 18 18 18 18 18 18							

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Ć1.00 Clinkers C1.05

Ċ1.10

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Ċ0.90

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Fig. 3 Quantitative and theoretical amount of  $C_4A_3S$  and the differences between them for different FR-CSA clinkers

3.1.3 Analysis of mineral compositions by FT-IR

3 To verify the variation of the mineral composition among the FR-CSA clinkers, FT-IR was used 4 to measure the chemical bonds of different phases in the FR-CSA clinkers. Infrared absorption spectra 5 of  $C_{0.90}$ ,  $C_{0.95}$ ,  $C_{1.00}$ ,  $C_{1.05}$ , and  $C_{1.10}$  are shown in Fig. 4. The vibration frequencies of the clinkers are in 6 the range of 400–1400 cm<sup>-1</sup>. As shown in Fig. 4, the absorption bands corresponding to  $C_4A_{3-x}F_x\overline{S}$ , 7 CaSO<sub>4</sub>, and C<sub>2</sub>S are intense, while the bands of C<sub>2</sub>AS, C<sub>5</sub>S<sub>2</sub> $\overline{S}$ , and C<sub>2</sub>F are hardly observed due to their 8 low content in all the clinkers. The appearance and increase in the intensities of the IR bands located at 9 about 595 cm<sup>-1</sup> result from the vibrations of Fe–O bonds in [FeO<sub>4</sub>] tetrahedral structural units[28]. The 10 bending vibration of the  $[SO_4]$  tetrahedron in CaSO<sub>4</sub> is identified at 617 cm<sup>-1</sup> and the stretching vibration of S-O for the [SO<sub>4</sub>] groups is observed at 1156 and 1195 cm<sup>-1</sup>. The enhancement of these 11 bands is observed in the clinkers with a  $C_m$  ranging from 0.90 to 1.10, which is consistent with the 12 13 results from XRD analysis. The groups of [SO<sub>4</sub>] in C<sub>4</sub>A<sub>3-x</sub> $F_x\bar{S}$  are identified at 619, 663, and 987 cm<sup>-1</sup>, 14 which are almost identical to the wavenumbers of the bands corresponding to  $[SO_4]$  groups in the 15 CaSO<sub>4</sub>[29, 30]. However, the bands observed at 1100 cm<sup>-1</sup> exhibit a shift to a higher wavenumber with 16 the addition of CaO in raw materials. This is because there are differences in the coordination 17 environment of the [SO<sub>4</sub>] group due to the incorporation of Fe<sub>2</sub>O<sub>3</sub> in C<sub>4</sub>A<sub>3-x</sub>F<sub>x</sub> $\bar{S}$ . The absorption bands 18 located at 411, 644, 690, 821, and 875 cm<sup>-1</sup> are assigned to [AlO<sub>4</sub>] groups in the C<sub>4</sub>A<sub>3-r</sub>F<sub>r</sub> $\overline{S}$  mineral[15, 19 31]. With the increase of the CaO content in raw material, the band observed at 821 cm<sup>-1</sup> shifted to 20 smaller wavenumber due to the decrease of Fe/Al in the  $C_4A_{3-x}F_x\overline{S}$  mineral. All of these phenomena are





22 23

Fig. 4 FT-IR of clinkers C<sub>0.90</sub>-C<sub>1.10</sub>

25 The SEM images of clinkers with a  $C_m$  ranging from 0.90 to 1.10 are presented in Fig. 5. The

- As shown in Fig. 5, with increasing CaO content in raw material, the amount of hexagonal platy grains
- 28 decreased slightly, which is consistent with the amount of  $C_4A_{3-x}F_x\overline{S}$  from XRD analysis.

<sup>26</sup>  $C_4A_{3-x}F_x\overline{S}$  exhibits a hexagonal platy structure or a quadrilateral columnar structure on the micro-level.



Fig. 5 SEM images of clinkers C<sub>0.90</sub>-C<sub>1.10</sub>

3 To verify the proportion of Fe<sub>2</sub>O<sub>3</sub> in the C<sub>4</sub>A<sub>3-x</sub>F<sub>x</sub> $\overline{S}$ , the elemental compositions of the C<sub>4</sub>A<sub>3-x</sub>F<sub>x</sub> $\overline{S}$ 4 grains are detected via SEM/EDS analysis. Five different  $C_4A_{3-x}F_x\overline{S}$  grains in clinker  $C_{1.00}$  are selected 5 to detect the elemental composition. The results are summarized in Fig. 6 and Table 7. The main 6 elements (i.e., O, Ca, Al, Fe, and S) are presented in the mapping picture, and their mass fraction are 7 listed in Table 7. The mass ratio of S: Ca: O is about 1: 5: 8, which is the chemical composition of 8  $C_4A_3\overline{S}$ . Furthermore, the total content of Fe<sub>2</sub>O<sub>3</sub> mixed with Al<sub>2</sub>O<sub>3</sub> are various in different grains, but the 9 average Fe<sub>2</sub>O<sub>3</sub> content in C<sub>4</sub>A<sub>3-x</sub> $F_x\bar{S}$  grains is approximately 14.08 wt.% for clinker C<sub>1.00</sub>. It is consistent 10 with the result calculated via quantitative XRD analysis.

Similarity, another four clinkers are also tested via SEM/EDS according to this method. The results are also presented in Table 7. The incorporation levels of  $Fe_2O_3$  in  $C_4A_{3-x}F_x\bar{S}$  decrease with increasing CaO content in raw material. Furthermore, the incorporation levels of  $Fe_2O_3$  in  $C_4A_{3-x}F_x\bar{S}$ calculated by two methods are consistent. The results further demonstrate that the content of  $Fe_2O_3$ incorporated in  $C_4A_{3-x}F_x\bar{S}$  decreased with increasing CaO content in raw materials.

16

Table 7 Chemical composition of areas in the  $C_4A_{3,x}F_xS$  grains (wt.%)

	Fusite $\gamma$ chemical composition of allows in the $C_{41}$ $S_{54}$ $p$ grams $(\gamma \alpha \gamma \beta)$								
Clinkers	Areas	0	Ca	Al	Fe	S	$W(Fe_2O_3)/W(Al_2O_{3^+}Fe_2O_3)$		
	1	41.73	24.03	22.38	4.47	5.23	13.12		
	2	42.08	24.83	19.93	5.29	5.46	16.71		
C <sub>1.00</sub>	3	42.34	24.14	21.86	4.14	5.27	12.52		
	4	41.94	22.66	22.31	5.04	5.92	14.59		
	5	42.49	24.68	21.63	4.43	5.18	13.41		
C <sub>0.90</sub>	Average	42.21	24.30	20.40	5.55	5.32	17.06		
C <sub>0.95</sub>	Average	42.39	24.16	20.96	4.64	5.69	14.34		
C <sub>1.00</sub>	Average	42.11	24.06	21.62	4.67	5.42	14.04		
C <sub>1.05</sub>	Average	43.44	23.88	21.94	3.88	5.53	11.79		
C <sub>1.10</sub>	Average	43.66	22.95	22.14	3.14	6.18	9.68		





3 3.1.5 Effective utilization rate of Fe<sub>2</sub>O<sub>3</sub> and Al<sub>2</sub>O<sub>3</sub>

4 As shown in Fig. 7, the ratios of  $Fe_2O_3$  and  $Al_2O_3$  distributed in different phases are presented 5 respectively. With the increase of CaO content in raw material, fewer Fe<sub>2</sub>O<sub>3</sub> participate in the formation 6 of  $C_4A_{3,x}F_x\overline{S}$ . Compared with the clinker  $C_{0.90}$ , about 50 wt.% Fe<sub>2</sub>O<sub>3</sub> transfer into C<sub>4</sub>AF and C<sub>2</sub>F from 7  $C_4A_{3x}F_x\bar{S}$  in clinker  $C_{1,10}$ . Because the hydration activity of  $C_4A_{3x}F_x\bar{S}$  is more beneficial to the 8 performance of the CSA cement based binders than that of C4AF and C2F[23], increasing CaO is not 9 conducive to the effective utilization of Fe<sub>2</sub>O<sub>3</sub>. For the distribution of Al<sub>2</sub>O<sub>3</sub>, although the amount of 10  $C_4A_{3x}F_x\bar{S}$  decreases obviously, the ratios of  $Al_2O_3$  distributed in them are almost unchanged, and all of them can reach to 90 wt.%. With the increasing CaO content in raw material, the Al<sub>2</sub>O<sub>3</sub> content in the 11 12 form of  $C_2AS$  decreases, while  $Al_2O_3$  content in the form of  $C_4AF$  increases. It is beneficial to the 13 mechanical property development of FR-CSA cement because C<sub>4</sub>AF has higher hydration reactivity 14 than C<sub>2</sub>AS. Therefore, increasing the amount of CaO in raw material can improve the effective 15 utilization of Al<sub>2</sub>O<sub>3</sub>, but it inhibits Fe<sub>2</sub>O<sub>3</sub> incorporation in C<sub>4</sub>A<sub>3-x</sub> $F_x\overline{S}$ .





1 2

Fig. 7 Ratios of Fe<sub>2</sub>O<sub>3</sub> and Al<sub>2</sub>O<sub>3</sub> distributed in different minerals (a: Fe<sub>2</sub>O<sub>3</sub>, b: Al<sub>2</sub>O<sub>3</sub>)



- 19 The effect of CaO on the phase composition of FR-CSA clinker is investigated based on FR-CSA
- 20 clinker with 30 wt.% C4AF. The XRD patterns of these clinkers are shown in Fig. 8. The types of

- 1 minerals in clinkers are almost the same with those in the FR-CSA clinker with 20 wt.% C<sub>4</sub>AF. For the
- 2 FR-CSA clinker with 30 wt.%  $C_4AF$  and a  $C_m$  of 0.90, the XRD peaks of  $CaSO_4$  are detected. The
- 3 characteristic peaks of C<sub>2</sub>AS don't exist in the FR-CSA clinker. More importantly, with increasing CaO
- 4 content in raw materials, the  $C_4AF$  content increases clearly and the main characteristic peak of  $C_4A_3$ .
- 5  $_{x}F_{x}\overline{S}$  also shifts to higher angles. It conclude that the incorporation content of Fe<sub>2</sub>O<sub>3</sub> in C<sub>4</sub>A<sub>3-x</sub>F<sub>x</sub> $\overline{S}$  also
- 6 decrease with increasing of CaO content in raw material, which is consistent with the phenomenon that
- $7 \qquad 20 \text{ wt.\% } C_4 AF \text{ is designed in the FR-CSA clinker.}$



Fig. 8 XRD patterns of the FR-CSA clinkers with 30 wt.% C4AF content designed

10 Thus, during the calcination process of FR-CSA clinker, the CaO content in raw material 11 influence the phase composition of clinker. With the increase of CaO content, the incorporation level of 12 Fe<sub>2</sub>O<sub>3</sub> in C<sub>4</sub>A<sub>3-x</sub>F<sub>x</sub> $\bar{S}$  decreases, which led to more C<sub>4</sub>AF and fewer C<sub>4</sub>A<sub>3-x</sub>F<sub>x</sub> $\bar{S}$  formed.

13

## 14 3.3 Effect of CaO in the form of CaSO<sub>4</sub> on iron-bearing phase formation of FR-CSA clinker

15 Fig. 9 presents the XRD patterns of FR-CSA clinkers designed with different extra amounts of 16 CaSO<sub>4</sub>, respectively. It demonstrates that a part of CaO in the form of CaSO<sub>4</sub> doesn't react with other 17 materials during the calcination process. In the proportion design of the raw mixture, all of CaO in the 18 form of  $CaSO_4$  are used to calculate the value of  $C_m$ . However,  $CaSO_4$  doesn't fully decompose for the 19 clinkers with the extra amount of  $CaSO_4$  more than 5 wt.%. Thus the  $C_m$  values are actually less than 20 1.00. Moreover, when the extra amount of  $CaSO_4$  of clinkers increases from 0 to 20 wt.%, the amount 21 of undecomposed CaSO<sub>4</sub> also increases. That is to say, the amount of CaO involved in the calcination 22 reaction decreases, which restricts the reaction of C2AS consumption according to Eq. 5, the amount of 23 C<sub>2</sub>AS increases in FR-CSA clinkers.

In addition, with increasing exctra CaSO<sub>4</sub> content in raw materials, the C<sub>4</sub>AF content in clinkers decreases, while the differences between the theoretically calculated and experimentally determined content of C<sub>4</sub>A<sub>3-x</sub>F<sub>x</sub> $\bar{S}$  increase. Meanwhile, the highest peak of C<sub>4</sub>A<sub>3-x</sub>F<sub>x</sub> $\bar{S}$  also shifts to lower angles according to their XRD patterns and the d-spaces of them increase from 3.7520 to 3.7614 Å. It demonstrates that the volume of the unit cell structure of C<sub>4</sub>A<sub>3-x</sub>F<sub>x</sub> $\bar{S}$  increases. Thus, according to the change of unit cell structure and the variation of mineral composition, it is concluded that the content of Fa O<sub>1</sub> increased in C A<sub>1</sub> =  $\bar{S}$  increases with increasing avtra CaSO<sub>2</sub> content in raw material

30 Fe<sub>2</sub>O<sub>3</sub> incorporated in C<sub>4</sub>A<sub>3-x</sub> $F_x\overline{S}$  increases with increasing extra CaSO<sub>4</sub> content in raw material





Fig. 9 XRD patterns of the FR-CSA clinkers with different CaSO<sub>4</sub> content in raw materials

In a word, with increasing extra  $CaSO_4$  content in raw materials, the CaO content involved in chemical reaction during calcination process decreases, leading to the increase of Fe<sub>2</sub>O<sub>3</sub> content incorporated in C<sub>4</sub>A<sub>3-x</sub>F<sub>x</sub>S. It is consistent with the variation caused by changes in CaO content.

3.4 Effect of CaO content variation caused by calcination temperature and holding time on the iron-bearing phase formation of FR-CSA clinker

9 Calcination temperature and holding time during calcination process of the FR-CSA clinker not 10 only affect the formation of reactive minerals but also can affect the decomposition of CaSO<sub>4</sub>. Thus the 11 effects of calcination temperature and holding time on the iron-bearing phase composition of the FR-12 CSA clinker were also investigated. The XRD results are shown in Fig. 10.

13 With the increase of calcination temperature, the intensity of CaSO<sub>4</sub> diffraction peaks decrease and 14 disappear finally. It means that more CaSO<sub>4</sub> were decomposed, resulting in that more CaO participate 15 in the mineral formation during the calcination process. Meanwhile, the intensity of C4AF diffraction 16 peaks increases, while the intensity of  $C_4A_{3,x}F_x\overline{S}$  decreases. It demonstrates that the content of Fe<sub>2</sub>O<sub>3</sub> 17 incorporated in the  $C_4A_{3-x}F_x\overline{S}$  decreases with the increase in calcination temperature. Similarly, as 18 shown in Fig. 10(b), with the increase in holding time, the intensity of  $CaSO_4$  diffraction peaks 19 decrease and those of C<sub>4</sub>AF increase. It is consistent with the phenomenon caused by the changing of 20 calcination temperature. Increasing the calcination temperature and holding time cause more CaO to 21 participate in the mineral formation reaction, accompanied by an increase in C<sub>4</sub>AF content and a 22 decrease in  $C_4A_{3-x}F_x\bar{S}$  content, which further demonstrates that more CaO content involved in the 23 reaction can make less Fe<sub>2</sub>O<sub>3</sub> incorporated in C<sub>4</sub>A<sub>3-x</sub>F<sub>x</sub> $\overline{S}$ . Therefore, when the calcination temperature is 24 between 1200 °C to 1300 °C, the effects of calcination temperature and holding time on the iron-25 bearing phase composition are mainly caused by their influence on the decomposition degree of  $CaSO_4$ 26 and the content of CaO involved in mineral formation reactions.



3

4 5

Fig. 10 XRD patterns of the FR-CSA clinkers with different calcination temperature and maintained time (a: Different temperature, b: Different holding time)

#### 4. Conclusion

6 This research demonstrated the effect of CaO content in raw material on iron-bearing phase 7 composition of FR-CSA clinker. The CaO content in raw material significantly influences the phase 8 formation in the FR-CSA clinker during the calcination process, especially for the formation of iron-9 bearing phases.

10 With the reduction of CaO content in the form of CaCO<sub>3</sub> or Ca(OH)<sub>2</sub> in raw materials, the amount 11 of C<sub>4</sub>AF formed in the FR-CSA clinkers decreased and the amount of Fe<sub>2</sub>O<sub>3</sub> incorporated in C<sub>4</sub>A<sub>3-x</sub>F<sub>x</sub> $\bar{S}$ 12 increased. When the C<sub>m</sub> is 0.90 and 20 wt.% C<sub>4</sub>AF are designed in the FR-CSA clinker, the value of *x* 13 in the C<sub>4</sub>A<sub>3-x</sub>F<sub>x</sub> $\bar{S}$  can be up to 0.36.

For the CaO content existed in the form of  $CaSO_4$  in raw material, it presents a similar effect. When the CaO content is constant and the  $CaSO_4$  content increases in raw material, the CaO content involved in the calcination reaction decreased, which also lead to an increase in the amount of  $Fe_2O_3$ incorporated in  $C_4A_{3-x}F_x\overline{S}$ .

The findings in this study make it possible to optimize the mineral composition of the FR-CSA clinker by changing the CaO content in the raw materials. Furthermore, low CaO content in raw material is beneficial to the formation of  $C_4A_{3-x}F_x\overline{S}$ , which enables that some solid wastes containing low calcium or low aluminum can also be used as raw materials for FR-CSA cement production.

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28 References

29 [1] Zhang L, Su MZ, Wang YM. Development of the use of sulfo- and ferroaluminate cements in

30 China. Advances In Cement Research. 1999;11(1):15-21.

- 31 [2] Glasser FP, Zhang L. High-performance cement matrices based on calcium sulfoaluminate-belite
- 32 compositions. Cement And Concrete Research. 2001;31(12):1881-6.

- 1 [3] Chen M, Liu B, Li L, Cao L, Huang Y, Wang S, et al. Rheological parameters, thixotropy and creep
- 2 of 3D-printed calcium sulfoaluminate cement composites modified by bentonite. Composites Part B:
- 3 Engineering. 2020:107821.
- [4] Chen IA, Juenger MCG. Synthesis and hydration of calcium sulfoaluminate-belite cements with
   varied phase compositions. Journal of Materials Science. 2010;46(8):2568-77.
- 6 [5] Liu C, Luo J, Li Q, Gao S, Su D, Zhang J, et al. Calcination of green high-belite sulphoaluminate
- 7 cement (GHSC) and performance optimizations of GHSC-based foamed concrete. Materials & Design.
  8 2019;182:107986.
- 9 [6] Hanein T, Galvez-Martos J-L, Bannerman MN. Carbon footprint of calcium sulfoaluminate clinker
- 10 production. Journal of Cleaner Production. 2018;172:2278-87.
- 11 [7] Hargis CW, Kirchheim AP, Monteiro PJM, Gartner EM. Early age hydration of calcium
- 12 sulfoaluminate (synthetic ye'elimite, ) in the presence of gypsum and varying amounts of calcium
- 13 hydroxide. Cement and Concrete Research. 2013;48:105-15.
- 14 [8] Morin V, Termkhajornkit P, Huet B, Pham G. Impact of quantity of anhydrite, water to binder ratio,
- 15 fineness on kinetics and phase assemblage of belite-ye'elimite-ferrite cement. Cement and Concrete16 Research. 2017;99:8-17.
- 17 [9] Wang YM, Su M, Zhang L. Sulphoaluminate Cement: Press Beijing Univ.
- 18 [10] Chen D, Feng XJ, Long SZ. The influence of ferric oxide on the properties of19 3CaO 3Al2O3 CaSO4. Thermochimica Acta. 1993;215:157-69.
- 20 [11] Möschner G, Lothenbach B, Winnefeld F, Ulrich A, Figi R, Kretzschmar R. Solid solution
- between Al-ettringite and Fe-ettringite (Ca6[Al1-xFex(OH)6]2(SO4)3·26H2O). Cement and Concrete
  Research. 2009;39(6):482-9.
- [12] Ectors D, Neubauer J, Goetz-Neunhoeffer F. The hydration of synthetic brownmillerite in
   presence of low Ca-sulfate content and calcite monitored by quantitative in-situ-XRD and heat flow
   calorimetry. Cement and Concrete Research. 2013;54:61-8.
- 26 [13] Cuesta A, Santacruz I, Sanf dix SG, Fauth F, Aranda MAG, De la Torre AG. Hydration of C4AF
- in the presence of other phases: A synchrotron X-ray powder diffraction study. Construction andBuilding Materials. 2015;101:818-27.
- [14] Huang Y, Shen XD, Ma SH, Chen L, Q. ZB. Effect of Fe2O3 on the formation of calcium
   sulphoaluminate mineral. Journal of the Chinese Ceramic Society. 2005;35(4):485-9.
- [15] Idrissi M, Diouri A, Damidot D, Greneche JM, Talbi MA, Taibi M. Characterisation of iron
   inclusion during the formation of calcium sulfoaluminate phase. Cement and Concrete Research.
   2010;40(8):1314-9.
- 34 [16] Touzo B, Scrivener KL, Glasser FP. Phase compositions and equilibria in the CaO-Al2O3-
- Fe2O3–SO3 system, for assemblages containing ye'elimite and ferrite Ca2(Al,Fe)O5. Cement and Concrete Research. 2013;54:77-86.
- 37 [17] Berrio A, Rodriguez C, Tobón JI. Effect of Al2O3/SiO2 ratio on ye'elimite production on CSA
- 38 cement. Construction and Building Materials. 2018;168:512-21.
- 39 [18] Ren C, Wang W, Li G. Preparation of high-performance cementitious materials from industrial
- 40 solid waste. Construction and Building Materials. 2017;152:39-47.

- 1 [19] Shen Y, Qian J, Huang Y, Yang D. Synthesis of belite sulfoaluminate-ternesite cements with
- 2 phosphogypsum. Cement and Concrete Composites. 2015;63:67-75.
- 3 [20] Yao X, Wang W, Liu M, Yao Y, Wu S. Synergistic use of industrial solid waste mixtures to
- 4 prepare ready-to-use lightweight porous concrete. Journal of Cleaner Production. 2019;211:1034-43.
- 5 [21] Wu S, Wang W, Ren C, Yao X, Yao Y, Zhang Q, et al. Calcination of calcium sulphoaluminate
- 6 cement using flue gas desulfurization gypsum as whole calcium oxide source. Construction and
- 7 Building Materials. 2019;228:116676.
- 8 [22] O. Andac FPG. Polymorphism of calcium sulphoaluminate (Ca4A160 16 S03) and its solid
  9 solutions. Advances in Cement Research. 1994;22(6):57-60.
- 10 [23] Bullerjahn F, Schmitt D, Ben Haha M. Effect of raw mix design and of clinkering process on the
- 11 formation and mineralogical composition of (ternesite) belite calcium sulphoaluminate ferrite clinker.
- 12 Cement and Concrete Research. 2014;59:87-95.
- 13 [24] Álvarez-Pinazo G, Cuesta A, Garc á-Mat é M, Santacruz I, Losilla ER, la Torre AGD, et al.
- 14 Rietveld quantitative phase analysis of Yeelimite-containing cements. Cement and Concrete Research.
- 15 2012;42(7):960-71.
- 16 [25] Cuberos AJ, De la Torre AG, Alvarez-Pinazo G, Martin-Sedeno MC, Schollbach K, Pollmann H,
- et al. Active iron-rich belite sulfoaluminate cements: clinkering and hydration. Environmental science
  & technology. 2010;44(17):6855-62.
- 19 [26] Sahu S, Majling J. Phase compatibility in the system CaO-SiO2-Al2O3-Fe2O3-SO3 referred to
- 20 sulphoaluminate belite cement clinker. Cement and Concrete Research. 1993;23(6):1331-9.
- 21 [27] Yao X, Yang S, Huang Y, Wu S, Yao Y, Wang W. Effect of CaSO4 batching in raw material on
- 22 the iron-bearing mineral transition of ferric-rich sulfoaluminate cement. Construction and Building
- 23 Materials. 2020;250:118783.
- [28] Rada S, Dehelean A, Culea E. FTIR, Raman, and UV-Vis spectroscopic and DFT investigations of
   the structure of iron-lead-tellurate glasses. J Mol Model. 2011;17(8):2103-11.
- 26 [29] Montes M, Pato E, Carmona-Quiroga PM, Blanco-Varela MT. Can calcium aluminates activate
- ternesite hydration? Cement and Concrete Research. 2018;103:204-15.
- 28 [30] Julphunthong P. Synthesizing of calcium sulfoaluminate-belite (CSAB) cements from industrial
- waste materials. Materials Today: Proceedings. 2018;5(7):14933-8.
- 30 [31] Guo Y, Deng J, Su M, Wang Y. A study on formation mechnism of ferrite phase in ferroaluminate
- 31 cement. Journal of the Chinese Ceramic Society. 1988;16(6):481-8.
- 32